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Diamond power devices: State of the art, modelling and figures of merit

To cite this article before publication: Nazareno Donato *et al* 2019 *J. Phys. D: Appl. Phys.* in press <https://doi.org/10.1088/1361-6463/ab4eab>

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Title

Diamond Power Devices: State of the Art, Modelling, Figures of Merit and Future perspective

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Table of Contents

Keywords.....

List of symbols and acronyms

Abstract

1. Introduction.....

2. Material requirements and modelling

 2.1. Substrates and growth

 2.2. Doping and defects.....

 2.3. Device surface termination.....

 2.3.1. Oxygen termination

 2.3.2. Hydrogen termination.....

 2.4. Heterojunctions with diamond.....

 2.5. Bulk and surface mobility.. ..

3. Figure of merits: definition and discussion.....

 3.1 Limits of existing figures of merits and their application to diamond power devices.....

 3.2 Device surface and current rating.....

 3.3 Breakdown voltage

 3.4. Switching losses

 3.5. Other criteria and figures of merit.....

4. Diamond devices for power converters.....

 4.1 Overview of diamond device architectures

 4.1.1. Diodes.....

 4.1.2. Metal-Oxide-Semiconductor devices

4.1.2.1.	Deep depletion MOSFETs.....	
4.1.2.2.	Inversion MOSFETs	
4.1.4.	JFETs, MESFETs and Bipolar Transistors.....	
4.1.5.	2DHG FETs	
4.1.6.	Vacuum switches.....	
4.2.	Comparison of 2DHG and bulk transistors	
4.3	Leakage current in diamond devices.....	
4.4.	Diamond devices with field relief designs	
4.5.	Packaging, thermal management and reliability.....	
5.	Benchmark of diamond power devices..	
5.1.	System level benefits and challenges...	
5.2.	Power converters with diamond devices..	
5.3.	Unipolar vs bipolar diamond diode	
5.4.	Benchmarking of diamond FETs against 4H-SiC and GaN..	
5.4.1.	10kV range.....	
6.	Future perspective of diamond power devices	
6.1.	The roadmap of diamond.....	
6.2.	Conclusions.....	

Keywords

Diamond, power devices, wide bandgap semiconductors, figures of merit, power converters, packaging, modelling, diamond roadmap.

List of symbols and acronyms

(B)FOM, (BALIGA) figure of merit

(N)PT, (non) punch-through design

2DHG (2DEG), two-dimensional hole (electron) gas

A^* , Richardson constant,

BJT, bipolar junction transistor

BV, breakdown voltage

$C_{T(BV)}$, transition capacitance

C_{OSS} , output capacitance

- 1
- 2
- 3 CVD, chemical vapor deposition
- 4
- 5 d, region thickness
- 6
- 7 dV/dt (di/dt), time derivative of voltage (current)
- 8
- 9 E_c , critical electric field
- 10
- 11 $E_{co(va)}$, conduction/valence band minimum/maximum
- 12
- 13 E_G , bandgap
- 14
- 15 E_{on} (E_{off}), energies loss densities by the diode during the turn ON (turn OFF) transient of the diode
- 16
- 17 EMI, electromagnetic interference
- 18
- 19 f, the switching frequency
- 20
- 21 FET, Field effect transistor
- 22
- 23 FLP, Fermi Level Pinning
- 24
- 25 HEMT, High electron mobility transistor
- 26
- 27 HPHT, High pressure high temperature
- 28
- 29 IGBT, Insulated Gate bipolar transistor
- 30
- 31 J_F (J_R), forward (reverse) current density
- 32
- 33 JFET, junction FET
- 34
- 35 k, Boltzmann constant
- 36
- 37 L, region length
- 38
- 39 MESFET, Metal-semiconductor FET
- 40
- 41 MOS, Metal oxide semiconductor
- 42
- 43 MOSFET, MOS field effect transistor
- 44
- 45 n, p , electron/hole concentration
- 46
- 47 $N_{A,D}$, active acceptor/donor concentration
- 48
- 49 $N_{A0,D0}$ total acceptor/donor concentration
- 50
- 51 NEA(PEA), negative (positive) electron affinity
- 52
- 53 $N_{sheet_{2DHG}}$, charge sheet concentration of the 2DHG
- 54
- 55 NTC(PTC), negative (positive) temperature coefficient
- 56
- 57 $N_v(N_c)$, valence (conduction) density of states
- 58
- 59 P, total power dissipation
- 60
- P_{disp} , power dissipations due to displacement current
- P_{static} ($P_{dynamic}$), static (dynamic) power density components for a power diode

$P_{sw.on}$, turn ON power losses

q , electron charge

Q_g , gate charge

Q_{gd} , gate-drain charge

Q_{oss} , output charge

Q_s , charge density stored in the PIN diode

R_{ON} , ON state resistance

R_{ON_spec} , specific ON state resistance

R_p , specific on state resistance of the p-type region of the diode

R_s , sheet resistance

R_{th} , thermal resistance

S , active area

T , absolute temperature (K)

T_j , junction temperature

V_{bi} , built-in voltage

V_{bn} , barrier height between the p-type semiconductor and the Schottky metal

V_{bn} , barrier height between the p-type semiconductor and the Schottky metal

V_F (V_R), forward (reverse) voltage

WBG, wide bandgap

δ , duty cycle

ε , permittivity

ε_0 , vacuum permittivity

ε_r , relative permittivity

η , ideality factor of the diode

$\mu_{n,p}$, electron/hole mobility

ρ , region resistivity

τ , ambipolar lifetime

ABSTRACT

With its remarkable electro-thermal properties such as the highest known thermal conductivity ($\sim 22 \text{ W/cm} \cdot \text{K}$ at room temperature) of any material, high hole mobility ($> 2000 \text{ cm}^2/\text{V} \cdot \text{s}$), high critical electric field ($> 10 \text{ MV/cm}$), and large bandgap (5.47 eV), diamond has overwhelming advantages over

silicon and other wide bandgap semiconductors (WBG) for ultra-high- voltage and high temperature applications ($>3\text{kV}$ and $>450\text{ K}$, respectively). However, despite their tremendous potential, fabricated devices based on this material have not delivered yet the expected high-performance. The main reason behind this is the absence of shallow donor and acceptor species. The second reason is the lack of consistent physical models and design approaches specific to diamond-based devices that could significantly accelerate their development. The third reason is that the best performances of diamond devices are expected only when the highest electric field in reverse bias can be achieved, something that has not been widely obtained yet. In this context, high temperature operation and unique device structures based on the 2DHG formation represent two alternatives which could alleviate the issue of the incomplete ionization of dopant species. Nevertheless, ultra-high temperature operations and device parallelization could result in severe thermal management issues and affect the overall stability and long-term reliability. Additionally, problems connected to the reproducibility and the long-term stability of 2DHG based-devices still need to be resolved.

This review paper aims at addressing these issues by providing the power device research community with a detailed set of physical models, device designs and challenges associated to all the aspects of the diamond power device value chain, from the definition of figures of merits, the material growth and processing conditions, to packaging solutions and targeted applications. Finally, the paper will conclude with suggestions on how to design power converters with diamond devices and will provide the roadmap of diamond devices development for power electronics.

1. Introduction

The increasing demand for a low carbon and energy efficient society has raised the need for new technologies for power electronics applications. In this context, wide bandgap (WBG) and ultra-wide bandgap (UWBG) semiconductors have been researched to quantify their advantages in terms of efficiency, current density, thermal performance, radiation hardness, switching frequency, and form factor of the overall power system compared to Si devices and systems [1-3]. The mature technology and the best trade-off between performance and cost have been so far the key to success of Silicon-based power devices and circuits. Nevertheless, there exists a huge variety of applications in the medium- to high- power (e.g. automotive sector, satellite communications, high speed trains, mobile terminals) where Si-based devices reach their limit in terms of efficiency due to ON state and switching losses and poor thermal dissipation management. While GaN and 4H-SiC devices have been successfully commercialized and also demonstrated to outperform their Si-based counterparts [4, 5], diamond and other UWBG semiconductors still face a number of challenges which are hindering the full exploitation of their superior physical properties (see table 1).

Despite its challenges, diamond has distinctive advantages when compared to other UWG semiconductors, due to its high hole-electron mobility, critical electric field, the highest known thermal conductivity and the widest bandgap [6, 7]. It has also peculiar features such as electron emission from hydrogen terminated surfaces, hopping conduction and surface transfer doping on hydrogen terminated surfaces. Recent breakthroughs have demonstrated efficient chemical vapor deposition (CVD) doping techniques for both p-type and n-type dopant species and relatively large area high pressure high temperature (HPHT) and CVD substrates [8]. Nevertheless, substrates are still limited in terms of cost and availability, and the resistivity of diamond layers is affected by the partial ionization of the dopants. More specifically, the lack of shallow dopant species is the main reason behind the poor room temperature performance of bulk diamond devices (see fig.1). In spite of this, several devices with high ON state current (up to 10A [9]), fast switching performance [10] and high breakdown voltage ($>2\text{kV}$) without any field relief structure [11] have been manufactured. Although the future commercialization of such devices seems to be limited only to niche applications (mainly

high power, frequency and temperature), future optimization of substrate growth techniques and device fabrication steps could enable the use of diamond devices in a wider range of applications.

This topical review is organised as follows. Section 2 focuses on the specific techniques to improve the doping efficiency and control, the unique properties arising from surface termination, the heterojunction structures and the carrier mobility for diamond. Section 3 presents a thorough investigation of the figures of merit applied to the specific scenario of diamond power devices and introduces a more global approach which allows to compare different diamond FETs. Section 4 deals with a systematic review of the applications and current state-of-the-art of diamond. It also highlights the issues that still need to be addressed prior to commercialization. In sections 5, the system level benefits of diamond diodes and FETs and their potential use in power converters are addressed. Finally, a suggested roadmap to a market-ready diamond power technology concludes this paper in section 6.

Table 1. Material properties of silicon, WBG and UWBG semiconductors for power applications.

Material		Silicon	WBG				UWBG		
			4H-SiC	GaN		Ga ₂ O ₃	Diamond		AlN
Bandgap (eV)		1.1	3.3	3.4		4.9	5.5		6.1
Critical Electric Field(MV/cm)*		0.3	2.8	3.5		8	7.7-20		10
RT Mobility (cm ² /Vs)*	electron	1500	1000	2000 (2DEG)	>1000 (bulk)	300	1060		300
	hole	480	120	<100 (2DHG)	<200 (bulk)	--	2100 (Bulk)	<300 (2DHG)	14
Thermal Conductivity(W/mK)		150	370	100(on Si) 165(on Shapphire) 253 (on GaN)		11-27	2200 - 2400		253-319
Relative permittivity (a.u.)		11.8	9.8	9		9.9	5.5		8.5
Substrate diameter(inch)**		8-17.7	8	8		4	<1		2
Substrate Dislocations (per cm ⁻²)		<10	10 ²	10 ⁴		10 ⁴	10 ⁴ -10 ⁶		10 ⁴
Saturation velocity (x10 ⁷ cm/s)	electron	1	1.9	2.5		2	2.5		1.4
	hole	0.8	1.2	---		----	1.4		--
Built-in Voltage (V)***		0.6	2.8	2.9		----	4.9		--
n-type dopants		available	available	available		available	moderate		moderate
p-type dopants		available	available	available		Not available	Available		poor
Commercially available devices		MOSFETs IGBTs Diodes Thyristors BJTs	Diodes BJTs MOSFETs	HEMTs		--	---		--

* Critical electric field and mobility are assumed to be doping independent.

** Typical size.

*** Calculation assumed constant doping for both sides of the junction (10x10¹⁵cm⁻³), room temperature conditions and bandgap values which can be found in the table.

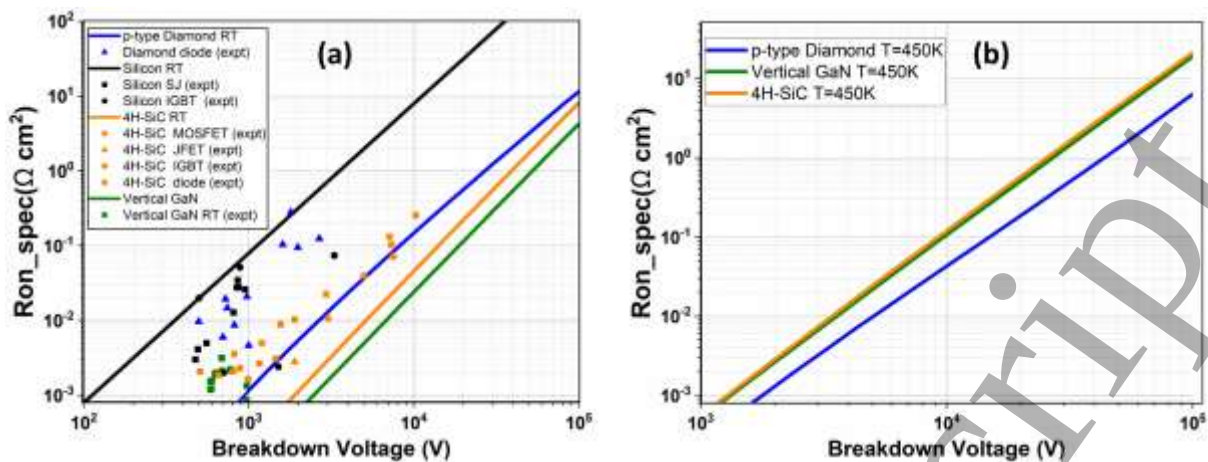


Figure 1, Vertical Ron_spec vs BV unipolar limit for semiconductors and comparison with experimental results at room temperature (T=300K) (a) and for high operating temperature (T=450K) (b). The calculated limit is the result of an optimization procedure which assumes PT profile for the electric field, mobility function of temperature and doping, temperature dependent breakdown field for 4H-SiC (calculated by means of the ionization integral) and the incomplete ionization in the case of p-type diamond. Silicon RT is used as reference considering a constant critical electric field (table 1). As it can be noted, boron doped diamond (p-type diamond) shows a better trade-off only for high temperature (b). Data taken from[11-21] and references therein.

2. Material requirements and modelling

2.1. Substrates and growth

Diamond crystals are usually classified on the basis of the type of impurities concentration (nitrogen and boron) and their arrangement in the crystalline structure. An accurate classification of diamond crystals can be found in table 2 and it applies to both natural and synthetic diamonds (HPHT or CVD).

Table 2. Classification of Diamond crystals based on the type and amount of impurities.

Diamond Substrates		
Type I It has enough nitrogen concentration (0.3%-0.5%) which can be measured with Infrared (IR) spectrometry.	Type Ia Nitrogen (N) atoms replace carbon (C) atoms in the lattice (N atoms are in substitutional lattice sites) and they tend to aggregate together.	Type IaA A specific type of Ia with N atoms pairs which occupy neighboring lattice site.
	Type Ib N atoms replace C atoms in the lattice, but they are isolated from each other. A great part of HPHT diamond substrates is type Ib.	Type IaB Cluster of 4 substitutional N atoms symmetrically surround a vacancy in the lattice structure.
	Type II It is characterized by a low nitrogen concentration which cannot be detected with IR. (usually <10 ¹⁷ cm ⁻³)	
	Type IIa Very low boron and nitrogen concentration which makes this form one of the purest diamond crystals available. Diamond gemstone can be included in this category.	
	Type IIb Boron concentration is higher than nitrogen. It has p-type semiconducting properties.	

The HPHT technique for the realization of synthetic diamond substrates allows to achieve high purity with a low defect density but the total size (between 2x2mm² up to 10x10mm² for IIa) is restricted due to intrinsic limitations of this method. HPHT substrates used for electronic devices are usually type Ib due to their relatively low cost and low dislocations density of about 10⁵cm⁻², but type IIa substrates can achieve even lower dislocations density (<10³cm⁻²) with drawbacks in terms of complex fabrication process and cost. CVD growth has less limitations on the size of the substrate (up to 0.5 inch) despite the fact that CVD does not allow to achieve the same crystalline quality of HPHT technique. Over 2

inches CVD substrates can be found in a mosaic configuration but the bonding boundaries between the wafers can limit the electrical performance of the devices and increase strain and defects in the structure [22]. An alternative technique to homoepitaxial growth is the heteroepitaxial growth of diamond on Iridium (Ir) and other similar substrates [23, 24]. This process allows to reach over 3 inches substrates but with a high dislocation density (between 10^7 - 10^9cm^{-2}).

2.2. Doping and defects

Due to the peculiar lattice structure and material strength, only shallow doping profiles ($<10\text{nm}$) can be obtained by means of high energy ion implantation process in diamond [25, 26]. Recently, thermal doping diffusion has been proven and a diamond pn diode based on this doping technique has been fabricated and characterized [27, 28]. However, these techniques require further investigations prior to becoming a reliable method of fabrication. Therefore, the incorporation of substitutional dopant species during the growth of diamond layers is mainly realized simultaneously with the CVD growth. Low boron concentrations (10^{15}cm^{-3}) are relatively easy to implement but fabrication of thick doped p-type layers remains challenging due to the loss of the crystallinity. While boron forms an acceptor level at 0.38eV from the maximum energy level of valence band (E_{va}), nitrogen and phosphorus n-type dopants result in a much deeper energy level from the conduction band minimum (1.7eV and 0.57eV from E_{co} , respectively). At the electro-thermal equilibrium, it is possible to solve the charge balance equation (with the Fermi statistic) to calculate the total number of holes (or electrons) for different temperature and compensation doping [29, 30]. This is shown in fig. 2(a-b). As it can be observed, compensation plays a key-role in the determination of the hole concentration. Besides, as discussed in [29], also the mobility and the overall resistivity of diamond layers are affected by the compensation level. Nevertheless, the substantial progress in the CVD growth of homoepitaxial grown have allowed for ultra-low compensated boron diamond layers ($<1\%$), even for low dopant density. Such compensation values which have been achieved by many research groups [31-33] are however still difficult to obtain with phosphorous doped layers. Therefore, as all the subsequent results and calculations shown in this manuscript will consider boron doped layers, compensation will be neglected. However, it is here worth mentioning that all the results can be easily extended in case of non-negligible dopant compensation.

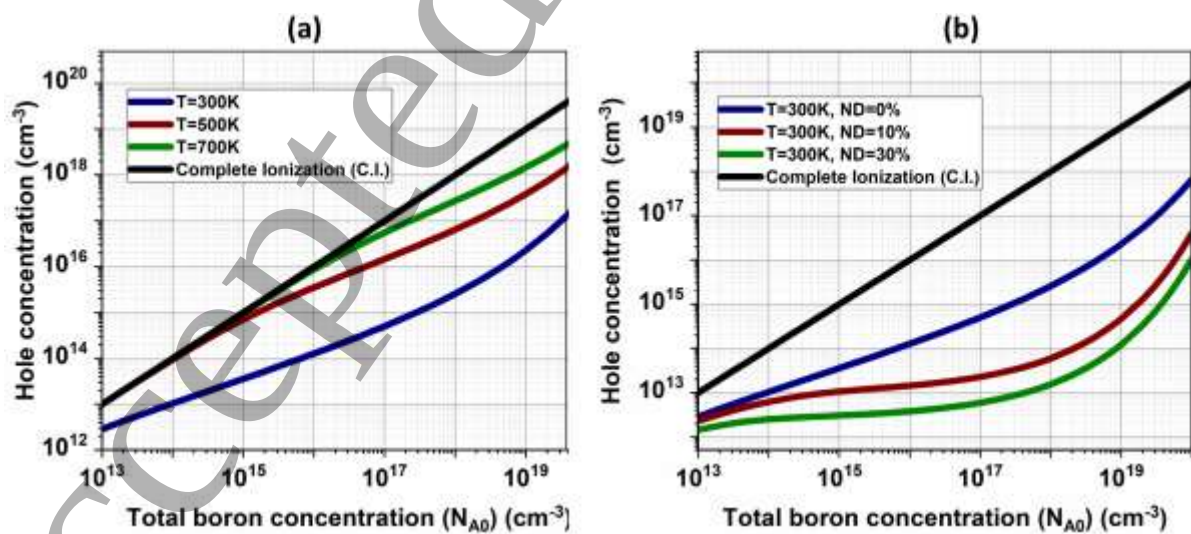


Figure 2, (a) hole concentration at the thermodynamic equilibrium vs total boron concentration (N_{A0}) for $T=300/500/700\text{K}$ and (b) effect of different donor compensation levels ($N_{\text{D}}=N_{\text{D0}}$) on the boron activation at $T=300\text{K}$. For the formulas used in the calculation, refer to [34, 35].

Growth of phosphorous doped diamond layer ensures the lower resistivity for n-type layers but it requires high and controlled temperature during the whole growth process [36]. Though it is possible to obtain a relatively wide doping window, heavy n-type ($>5 \times 10^{19} \text{cm}^{-3}$) doping still remains challenging [37]. In addition, the crystal orientation also plays a key role in determining the quality of the doped and intrinsic layers. The $\langle 100 \rangle$ orientation is the most common for growing diamond layers. However, it is still complicated to grow n-type layers and there are still limitations in the efficiency of p-type doping. Conversely, in the $\langle 111 \rangle$ direction n-type phosphorous dopants can be incorporated much easier and it is possible to achieve one of the highest boron concentration [38]. Nevertheless, one of the significant drawbacks of the $\langle 111 \rangle$ orientation is the formation of macroscopic defects which leads to a poorer quality of the material [39]. On $\langle 110 \rangle$ faces boron concentration can be improved if compared with $\langle 100 \rangle$ but the reduced surface area hampers the benefits due to the enhanced doping control. Other orientations like $\langle 113 \rangle$ which have not been deeply investigated yet, may result in enhanced control and speed for the doping process of diamond layers. Macroscopic and microscopical defects are also playing a key role in determining the electrical properties [40, 41]. Non-epitaxial crystallites, which are a typical feature of homoepitaxial grown diamond, have already been demonstrated to affect the performance of MESFETs, MOS and Schottky diamond diodes [42-45]. For a complete review of diamond defects and their characterization techniques, the reader can refer to [46].

Table 3. Available doping windows for the doping of diamond electronic devices. n-type doping refers to Phosphorous.

	Available		Under development / required	
	<i>min</i>	<i>max</i>	<i>min</i>	<i>max</i>
n-type doping concentration	$\approx 3 \times 10^{15} \text{ cm}^{-3}$ [47]	$\approx 8 \times 10^{19} \text{ cm}^{-3}$ [48]	$\approx 1 \times 10^{14} \text{ cm}^{-3}$	$> 1 \times 10^{20} \text{ cm}^{-3}$
n-type layer thickness	$< 100 \text{ nm}$ [47, 49]	$\approx 5 \mu\text{m}$ [36]	-----	$> 50 \mu\text{m}$
p-type doping concentration	$\approx 1 \times 10^{15} \text{ cm}^{-3}$ [50]	$> 1 \times 10^{21} \text{ cm}^{-3}$ [51]	$< 1 \times 10^{14} \text{ cm}^{-3}$	-----
p-type layer thickness	$< 10 \text{ nm}$	$\approx 100 \mu\text{m}$ [51]	-----	$> 200 \mu\text{m}$

2.3. Device surface termination

2.3.1. Oxygen termination

Oxygen termination is generally used to improve the adhesion of oxide and diamond layers and it induces a positive electron affinity (PEA) of 1.7eV. One of the main drawbacks of such passivation is the high FLP effects generated by the presence of high density interface states [52]. Ozone treatment and immersion in hot mixed acid are the most common treatments adopted by researchers to induce O-termination in diamond layers [53]. Oxygen terminated (O-terminated) diamond is also exploited for the removal of the hole-type conductive layer (2DHG) [11].

2.3.2. Hydrogen termination

On hydrogen terminated (H-terminated) diamond surfaces, both a negative electron affinity (NEA) of $> -1 \text{ eV}$ and a strong FLP are induced [54, 55]. Diamond H-terminated surfaces, which can be obtained by either hot filament or plasma treatment, have been widely explored due to their unique property of surface conductivity. Even though the origin of surface conductivity is still not well understood [56], the presence of adsorbates (i.e. materials with higher binding energy than H-diamond) on C-H diamond surface and the local exchange of electrons with the diamond valence band is the most likely

explanation for the formation of the 2DHG. One can also note that these properties have been also presented with polycrystalline diamond [57] or heteroepitaxial grown diamond [58].

2.4. Heterojunctions with diamond

Among the possibilities which allow to obtain a RT fully activated diamond channel, heterojunctions between diamond and group III nitrides (AlN, GaN and BN) are one of the most promising and attractive configurations. As the growth of GaN layers on diamond surfaces is complicated, AlN and BN have been identified as best materials for diamond heterojunctions. Kuech et al. [59] reported an H-terminated diamond surface with an AlN passivation layer and the first demonstration of a AlN/diamond heterojunction np diode was successfully carried out by Miskys et al. [60] by using a MBE (Molecular Beam Epitaxy) technique. As the H-terminated surface results in a poor attachment to the AlN layer, an O-terminated diamond surface was adopted for the first AlN/diamond heterojunction FET realized by Imura et al. [61, 62] with a metal organic vapor phase epitaxy technique.

2.5. Bulk and surface mobility

Discrepancies between time-resolved cyclotron resonance (TRCR), time-of-flight (TOF) and Hall measurements have generated confusion about the real value of diamond carrier mobility with overestimations for hole and electron mobility at RT ($7300\text{cm}^2/\text{Vs}$ for electron and $5300\text{cm}^2/\text{Vs}$ for holes [63]). More specifically, Hall electron mobility values calculated at RT oscillate around $1000\text{cm}^2/\text{Vs}$ whilst TOF performed by Isberg et al. [64] shows higher electron mobility, an overestimation which may be caused by the approximation of Hall scattering factor, as suggested by Pernot [65]. Regarding the electron mobility in n-type layers, intra-valley phonon scattering is the dominant scattering mechanism in the high temperature range (regardless of the doping level of the layer) while the interaction with the intra-valley acoustic phonon is the main scattering mechanism in the middle temperature range [65, 66]. In the low temperature range, ionized impurity and neutral impurity scattering are the main scattering mechanisms. Conversely, hole mobility is subjected to the same scattering mechanisms in the low temperature range, intra-band and inter-band acoustic phonon scattering are dominating the medium range and the interaction with the optical phonon is the main mechanism responsible for the mobility at high temperature (HT).

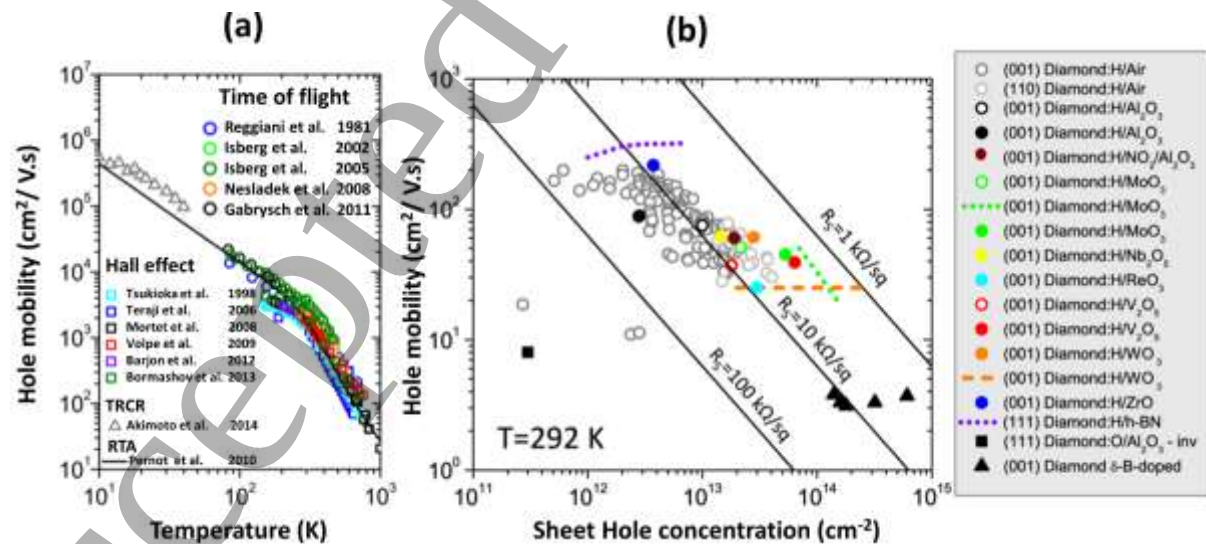


Figure 3, (a) Hole mobility vs temperature for bulk diamond (experimental and theoretical). The gray rectangle corresponds to typical room temperature 2DHG mobility in diamond FET, (b) Hole mobility for H-terminated diamond FETs, O-terminated diamond FETs and delta B-doped diamond. Data from [65] (references therein) and [67].

Regarding the hole mobility, some discrepancies between TOF, TRCR and Hall measurements still persist with values ranging between $3800\text{cm}^2/\text{Vs}$ and $2100\text{cm}^2/\text{Vs}$ at RT with a tendency of measurements to confirm the $2100\text{cm}^2/\text{Vs}$ value [65, 68] (figure 3(a)). However, recent measurements tend to agree on the RT values for electron/hole mobilities [31, 32].

Such mobility values can be reached in pure or low doped diamond, where the limiting mobility mechanism is purely intrinsic due to phonon scattering mechanisms. Detailed analysis of the mobility dependence versus doping level concerning phosphorus doped n-type [66] and boron doped p-type [29, 69, 70] materials have been reported. In uncompensated and highly doped material, the neutral impurity scattering is the dominant scattering mechanism because of the large ionization energy of the donor and acceptor dopants. In figure 4, the hole mobility has been plotted as function of the temperature-doping with the fitted models from [66, 70].

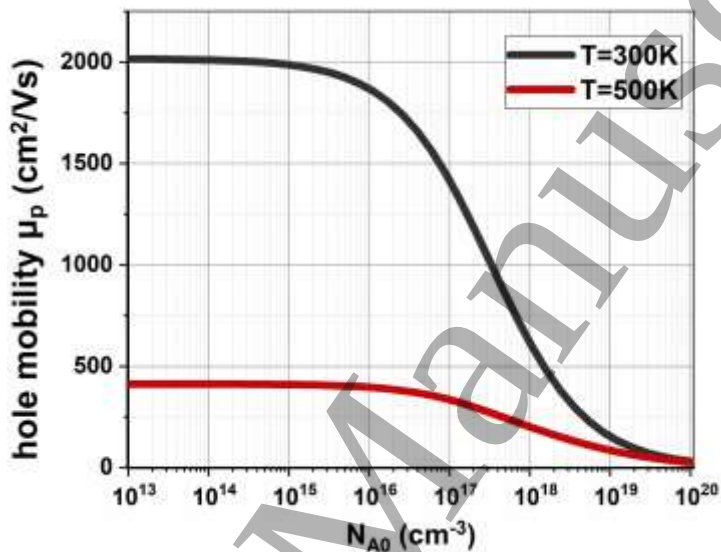


Figure 4, hole mobility vs doping for boron doped diamond for T=300/500K. The parameters and equations can be found in [65, 66, 70].

Few studies have analyzed the mobility in hydrogen terminated diamond surfaces. In general, extraction of the conductivity (carrier sheet density and mobility) is obtained during the electrical characterization of the 2DHG FETs. Besides, values of surface channel p-type channel FETs rarely exceed $200\text{--}300\text{cm}^2/\text{Vs}$ due to surface roughness, ionized impurity scattering, and the high surface electric field generated by the presence of the negatively charged acceptors which are causing the confinement of the 2D hole gas (figure 3(b)). Recently, Li et al. [71] calculated the 2DHG mobility as function of the temperature and the hole gas density and then compared their theoretical results with a variety of experiments.

Mobility extraction has been also performed on delta doped FETs showing that the predicted enhanced mobility in such layers cannot be achieved and values rarely overcome $20\text{cm}^2/\text{Vs}$ [72]. On C-OH diamond surfaces, the mobility of the inversion layer on lateral MOSFET have been estimated to be $8\text{cm}^2/\text{Vs}$ due to the non-optimal quality of the diamond/ Al_2O_3 oxide interface [73].

3. Figure of merits and system-level comparisons: definition and discussion

3.1 Limits of existing figures of merits

Power semiconductor devices are characterized by conduction, switching and OFF state losses. A perfect figure of merit would take each contribution into account, with specific interactions at the system level (i.e. thermal, driving, electromagnetic compatibility, reliability, sourcing and cost). Unfortunately, it is almost impossible to compare different devices based on different technologies and/or materials based on a simple figure of merit. As an example, switching losses are not only dependent on the power device itself but also on the driving circuit, the topology employed (e.g. based on soft or hard switching) and parasitics associated with packaging. One of the most used figure of merit in power semiconductor devices is Baliga's Figure of Merit (BFOM) defined in [74] and equation (1). This BFOM has been derived from the specific ON state resistance (equation 2), which can be expressed by equation 3 in the case of several assumptions. Consequently, equation 3 introduces the BFOM in the typical trade-off between the specific ON state resistance (R_{ON_spec}) and the breakdown voltage (BV). However, the assumptions required to directly relate the BFOM with the R_{ON_spec} cannot apply in the context of diamond power devices; in diamond bulk devices, the incomplete ionization of dopants, and in 2DHG devices, the sheet carrier concentration and specific 2DHG mobility must be considered. Consequently, equation (3) is no longer valid and the specific R_{ON_spec} is no longer derived by the BFOM. In equations (1-3), μ_n is the mobility for electrons (μ_p for holes), ϵ is the dielectric permittivity of diamond, E_c is the critical electric field (table 1), ρ_{Drift} is the resistivity, S is the active area, L_{Drift} is the length of the drift region and q is the electron charge.

$$BFOM = \mu_{n,p} \cdot \epsilon \cdot E_c^3 \quad (1)$$

$$(General\ case): R_{ON_spec} = R_{ON}S = \rho_{Drift} \cdot L_{Drift} = \frac{L_{Drift}}{q \cdot (\mu_{n,p} \cdot n, p)} \quad (2)$$

$$(With\ assumptions): R_{ON_spec} = \frac{4 \cdot BV^2}{\mu_{n,p} \cdot \epsilon \cdot E_c^3} = \frac{4 \cdot BV^2}{BFOM} \quad (3)$$

Consequently, the specific ON state resistance (R_{ON_spec}) is used as a figure of merit to compare different devices or materials, for a given range of breakdown voltages. The ON state resistance is typically measured by pulsed I-V or calculated based on analytical formula or numerical analyses. The device area is extracted from the active area or device area, including or not the termination region. The breakdown voltage is measured or calculated based on specific hypotheses. There are mainly four issues with the direct comparison of the R_{ON_spec} value among different devices or materials at the same breakdown voltage and the use of R_{ON_spec} as a figure of merit: the lack of direct switching loss estimation, the different assumptions related to R_{ON} , S and BV between devices or materials, the lack of link with the thermal conductivity and the scalability of R_{ON} with the surface. The junction temperature at which the comparison is carried out must also be discussed.

In order to relate the figure of merit to the switching losses, other figures of merit such as $R_{ON} \cdot Q_g$ (or Q_{gd} , Q_{oss}) have been introduced [1, 75, 76]. These figures of merit are clearly more complex than the R_{ON_spec} FOM alone, albeit harder to predict for diamond power devices. Indeed, actual diamond power devices still have small active areas which makes difficult a precise measurement of the capacitances related to the active area. These figures of merits are best suited for unipolar devices but cannot be used in the context of bipolar devices due to recovery charges and their impact on switching losses. Specific studies are required on diamond power devices optimization and measurements, to further demonstrate low Q_g (gate charge), Q_{gd} (gate-drain charge) and Q_{oss} (output charge), whereas

most of the recent achievements concentrated on reductions in R_{on_spec} . Accordingly, the control of the Miller ratio between Q_{gd} and Q_{gs} is also an important criterion to consider. Immunity to dV/dt and dI/dt and the maximum turn ON and turn OFF switching speeds are equally relevant. Besides these parameters, the gate leakage must also be considered.

3.2 Switching losses

While estimating the conduction loss is straightforward with appropriate conduction models, the prediction of switching losses highly depends on multiple parameters such as the parasitic capacitances, the gate driver parameters (e.g. transconductance, min and max gate voltage) and circuit elements (e.g. parasitic inductances and capacitances). A fair comparison for switching losses must include similar electromagnetic compatibility (EMC)/electromagnetic interference (EMI) constraints, as large gate currents in MOSFET will lead to reduced switching losses but very high dV/dt and dI/dt values. Such high transient values can have negative impacts on motors, cables, common mode filters [77], and can cause false switching through the Miller capacitance [78]. Moreover, to date there are no studies on the switching losses in diamond FETs, mainly due to the limited availability and small size of diamond FETs. A fair comparison of the switching losses among power devices based on different materials will depend on the specific application. There are however a few case studies of diamond diodes in power commutation cells [79-81], mainly on SBD diamond diodes showing small recovery-like currents due to the diode intrinsic transition capacitance. The main problem in performing these experiments is to associate small size diamond diodes with power FETs having similar voltage capability and parasitic capacitances. The small signal and large signal characterization of diamond FETs are then highly desired to be able to benchmark accurately the performances of diamond power devices [82, 83].

The expected benefits for power electronics with unipolar diamond or ultra-wide band gap devices are to be able to match the conduction loss of silicon bipolar devices. Given the unipolar conduction and the absence of excess charge in the ON state, the switching losses could be dramatically reduced. As presented in section 5.3, the benefits of bipolar diamond devices with an efficient resistivity modulation will be limited to ultra-high voltage and low switching frequency, due to the large built-in potential in diamond and short carrier lifetimes. Consequently, unipolar diamond devices are expected to have the highest impact at system level in the short- to mid-term. Despite the issues related to switching loss predictions with diamond power devices discussed hereinbefore, one can assume several hypotheses to predict the switching performances of diamond unipolar power devices; the turn OFF losses with diamond FETs will be neglected as the channel current is turned OFF very quickly thanks to the smaller input capacitance (smaller active area) and the high transconductance; the turn ON losses are not limited by EMI issues; the drift region is considered in NPT configuration; the device is of vertical type. As a result, the lowest possible switching losses in a power FET are governed by the stored electric charge in the output capacitance (C_{oss}) during the switching transition, where the C_{oss} as a function of V_{DS} can be expressed by equations (4) and (5). In equation (4), $C_{oss(V)}$ is the output capacitance as a function of the bias, which is typically the transition capacitance $C_{T(V)}$ exhibiting a square root dependence with bias when the drift region is in NPT condition. At the breakdown voltage, the transition capacitance $C_{T(BV)}$ is calculated by equation (5), with $\epsilon_0 \times \epsilon_r$ the permittivity of diamond, S the active area, d_{Drift} the thickness of the drift region. Whereas two FETs or one FET and one diode will be associated in a power commutation cell, the active area of each power device can be different as a function of the duty cycle. As a consequence, only the C_{oss} of one power device can be considered for the estimation of the minimum power losses $P_{sw.on}$,

as proposed for example in [84] and equation (6), where V is the switched voltage and f the switching frequency.

$$C_{OSS(V)} = C_{T(V)} = C_{T(BV)} \sqrt{\frac{BV}{V}} \quad (4)$$

$$C_{T(BV)} = \epsilon_0 \times \epsilon_r \times \frac{S}{d_{Drift}} = C_{T(BV)}^* \times S \quad (5)$$

$$P_{sw.on}(V) = \frac{2}{3} \times C_{T(BV)} \times \sqrt{BV} \times V^{\frac{3}{2}} \times f \quad (6)$$

3.3 Other criteria for system-level comparison

In an actual application the best power device is the one minimizing total losses while respecting key constraints (e.g. maximum junction temperature and power density). Therefore, the optimal device area minimizing the sum of switching losses (given by (6)) and conduction losses (related to (2) or (3)) can be determined for a fixed set of specifications (switching frequency, breakdown voltage, rated current, NPT profile) thanks to the models and discussions presented earlier in this section and in other articles such as [84]. Figure 5 shows an example of a 1.7kV diamond vertical unipolar power FET, switching 1200V-50A at 20kHz or 100kHz, with a maximum junction temperature of 425K/450K and a duty cycle of 0.5. At the optimal area of 0.25cm² (20kHz), the total losses represent 12W (0.04% of switched power), the current density in ON state is 200A/cm² and the total power loss density is 54W/cm² (similar results are observed between T=425K and 450K). If the switching frequency is increased up to 100kHz, the optimal area becomes 0.1cm², the total losses 27W, the current density in ON state 500A/cm² and the total power loss density 270W/cm² (figure 5(a)). If this power density is too high for the thermal spreader, the active area must be increased, in the price of increased total power losses, or the switching frequency must be reduced. This simple approach can be applied to compare different devices or materials for the same specifications. As an example, a commercially available 1700V SiC MOSFET has been chosen for reference [85] and while applying the same modelling, figure 5(b) presents the performances for both materials under the same specifications (20kHz switching frequency). The diamond device will be almost ten times smaller, with more than three times lower total losses than SiC. This is even done at a higher temperature of 450K for diamond comparatively to 425K for SiC, which is also a huge benefit at the system level (see section 5).

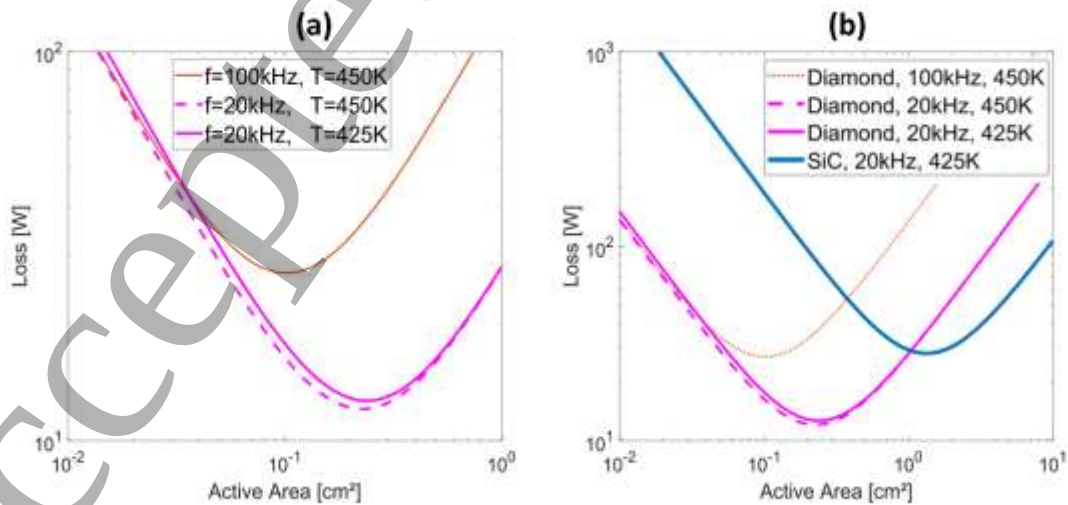


Figure 5, (a) Total losses as a function of active area for 1.7kV diamond vertical MOSFETs operating at 425K/450K junction temperature, switching 1.2kV, 50A at 20kHz or 100kHz, with a duty cycle of 0.5. Similar results are observed at T=425K and T=450K. At f=100kHz the two

curves $T=425\text{K}$ and $T=450\text{K}$ overlap (b). Total losses as a function of active area for a 1.7kV diamond vertical MOSFET operating at 425K/450K junction temperature, and a 1.7kV SiC vertical MOSFET operating at 425K, both switching 1.2kV, 50A at 20kHz, with a duty cycle (δ) of 0.5.

Considering that power loss density is still around or below $50\text{W}/\text{cm}^2$, the switching frequency can be increased up to 100kHz – see table 4 for the complete analysis. At both optimal design points, the total losses with diamond switching at 100kHz with a junction temperature of 450K are still smaller than the total losses of SiC switching at 20kHz. The power density has been however increased from $20\text{W}/\text{cm}^2$ in 20kHz SiC to $270\text{W}/\text{cm}^2$ with the 100kHz Diamond. Increasing the diamond area above the optimal area will increase the total losses, while slightly decreasing the power density. In this example, it is possible to reduce the power density in diamond from $270\text{W}/\text{cm}^2$ to $153\text{W}/\text{cm}^2$, while increasing the active area from the optimal value of 0.1cm^2 to 0.26cm^2 . At this “non-optimal” design (in terms of total loss), the total loss in diamond at 100kHz is the same as the SiC operating at 20kHz for the same switched current and voltage. To conclude this analysis, a global comparison can be proposed and is represented in figure 6: the performance of diamond power devices is highlighted here, while offering at the same time, smaller active areas, smaller total losses, larger current densities and higher junction temperatures. Table 5 shows the comparison of diamond and SiC under the same switching conditions at room temperature, where the benefits of diamond are reduced due to the high activation energy (no compensation is assumed). In spite of the benefits of diamond in terms of total losses and active area, the power loss density has been increased by 250%.

Table 4. Comparative case study between SiC and diamond for the same application.

1200V (BV 1700V) 50A 0.5 duty cycle		Diamond 20kHz	SiC 20kHz	Diamond 100kHz	SiC 100kHz	Diamond 100kHz* – *NON OPTIMAL
Optimal Area	cm^2	0.25	1.85	0.1	0.85	0.26
Conduction Loss	W	≈ 6	≈ 20	≈ 13.5	≈ 45	≈ 5.3
Switching loss	W	≈ 6	≈ 20	≈ 13.5	≈ 45	≈ 34.7
Total loss	W	12	40	27	90	40
Junction Temperature	K	450	425	450	425	450
Current density	A/cm^2	200	27	500	59	191
Power loss density	W/cm^2	54	20	270	105	153

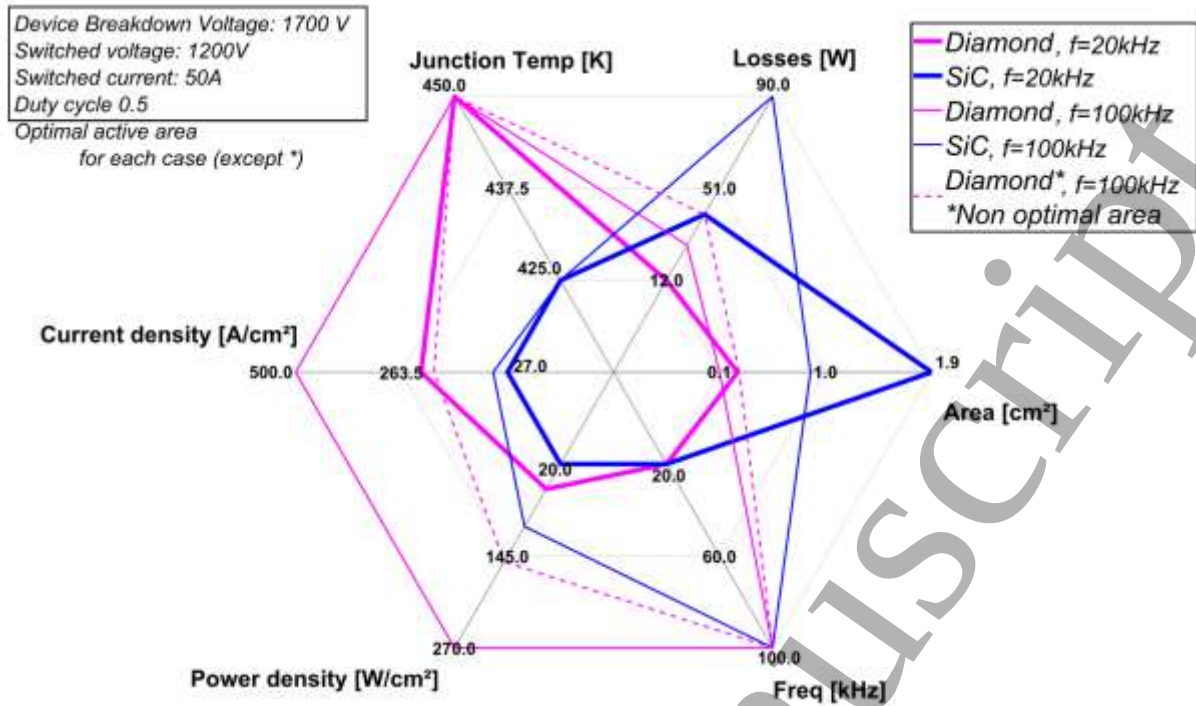


Figure 6, Spider chart comparing diamond and SiC (table 4 data).

Table 5. Comparative case study between SiC and diamond for the same application, at room temperature.

1200V (BV 1700V) 50A 0.5 duty cycle		Diamond 20kHz	SiC 20kHz
Optimal Area	cm ²	0.4	1.35
Conduction Loss	W	≈10.7	≈14
Switching loss	W	≈10.7	≈14
Total loss	W	21	28
Junction Temperature	K	300	300
Current density	A/cm ²	125	37
Power loss density	W/cm ²	53	21

Please note that in these comparisons, the following parameters for the SiC MOSFET have been taken from its datasheet: initial active area of 0.3cm² (overestimate of active area), with an ON state resistance of 100mΩ and 50mΩ, respectively at 425K and room temperature, 180pF parasitic output capacitance at 1kV. The performance of the SiC MOSFET has been linearly scaled and compared with diamond at the optimal area under the same operating conditions.

4. Diamond devices for power converters

4.1.1 Diodes

Due to the low incorporation of phosphorous and the high activation energies for n-type dopants, diamond Schottky diodes have been mainly fabricated on boron doped layers. Fewer benefits would be obtained with bipolar devices due the high built-in voltage of the pn junctions (table 1), which would result in a significant ON state voltage drop. The bipolar mode could only be of use in ultra-high voltage applications (above 10kV) and low-medium frequency applications, as it will be discussed in paragraph 5.3. Manufactured diodes have been reported both featuring unipolar action such as Schottky, Metal-Intrinsic-P (MIP), Schottky PN diode (SPND) and bipolar action such as PN junctions and PIN diodes (figure 4).

Regarding p-type diamond Schottky diodes, high blocking voltages (up to 10kV [86]) and critical electric field (7.7MV/cm [21]) have been reported in the literature but a significant non-uniformity in the material quality has resulted in discrepancies for the reported experiments. Record currents of several Ampere have been measured for a few packaged diodes [9] and high temperature operations (over 525K) have been experimentally demonstrated with no observed degradation of the Schottky properties [21, 42, 50, 53, 87-92]. The difficulties in obtaining large size of self-standing low-resistive single crystal have resulted in the development of pseudo-vertical diamond structures (figure 7), where the p++ layers, on which the ohmic contact is deposited, is grown on top of the HPHT substrate. Various metals (W, Zr, Cu, etc.) and surface treatments have been explored in order to optimize the rectification behavior [93-95], to improve the uniformity of the Schottky metal [42] and to reduce the leakage currents, which result in premature breakdown [53, 96, 97]. The best trade-off obtained so far has been achieved with Zr (table 6) [21] for pseudo-vertical diamond diodes, while a 5A, >1kV vertical diode has been demonstrated in [87] with Pt Schottky metal.

Diamond vertical PIN diodes have been successfully reported in [98-101]. Together with the lack of carrier lifetime control, reproducibility and uniformity, the high built-in voltage of the pn junction (even when operated at high temperature) represents a limiting factor in the development of this device structure [102]. To overcome some of the previously mentioned issues, Schottky pn type diodes have been suggested [103-106]. The Schottky metal on top of the n-type layer is able to deplete the n-type (nitrogen or phosphorous) doped layer in both the ON state and OFF state, allowing for holes to be injected from the p+ layer in the ON state and at the same time support the reverse voltage.

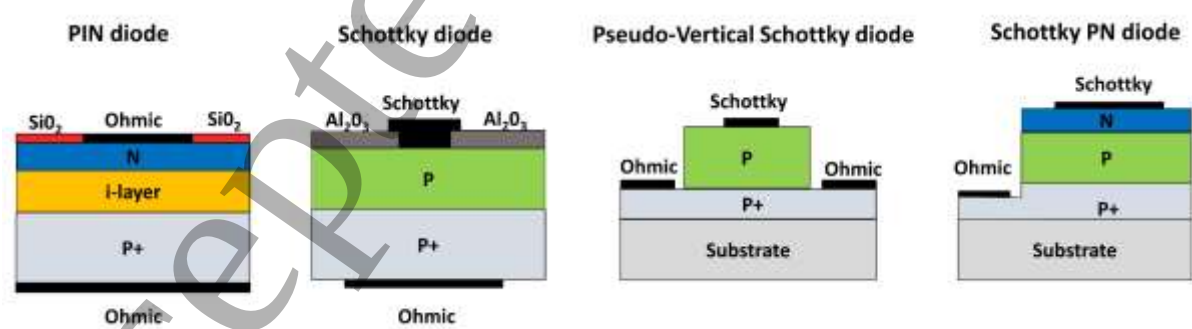


Figure 7, Different device structures for diamond diodes. From the left PIN diode, vertical Schottky, pseudo-vertical Schottky, SPND. Best values from different devices are reported in the table 6.

Table 6. State-of-the-art parameters and key features for diamond diodes depicted in figure 7. ON state current and current density has been extracted and reported for different bias conditions.

Device	PIN Diode	Vertical Schottky	Pseudo-vertical Schottky	Schottky PN diode
Conduction mode	Bipolar	Unipolar	Unipolar	Unipolar
ON state Current	<100mA at V=5V with T=300K [98, 99, 107]	20A at V=1.8V with T=300K [108] >20A at 1.2V with T=500K [108],	≈100mA at V=5V with T=300K [42, 109]	<100mA at V=7V with T=300K [106]
Breakdown voltage	>11kV [110]	>1.8kV at T=300K [111],	>1.6kV at T=300K [112]	>55V at T=300K [103]
Current Density	>100A/cm ² at V=30V <10A/cm ² at V=10V with T=300K [99] >100A/cm ² at V=10V with T=500K [98])	>100A/cm ² at V=2V with T=300K [113] ≈100A/cm ² at V=1.2V with T=500K [108]	< 100A/cm ² at V=2V 4500A/cm ² at V=7V with T=300K [21, 42] >200A/cm ² at V=2V (after Zr annealing at T=750K)	<10A/cm ² at V=2V >60kA/cm ² at V=6V with T=300K [103]
Notes	High built-in voltage. Need long lifetime for minority carriers (state-of-the-art value is estimated to be 6ns for holes [114]) and highly doped n+ region). Positive temperature coefficient of the BV [99]. Employed as slow neutrons detector [100].	When the drift region is lowly doped the device is known as MIP+ diode. MIP+ diode shows Space charge limited current behavior. High scalability and fast turn OFF (≈ns). BV limited by defects.	Low scalability of the BV and the Ron_spec. Highest dielectric field strength reported (7.7MV/cm). Schottky metal stable up to 700K. Etching of p+ is needed to avoid common substrate issues [115]. Used as temperature sensor [115].	No theoretical trade-off between BV and Ron_spec. Highest reported current density for diamond diodes. Positive temperature coefficient for the ON state current. High switching speed ≈10 ns for low reverse voltage (≈-5V) [104]. Thickness and doping of the n-type layer set a limit for the scalability of the device. Exploits hopping conduction mechanism for the p+ layer. Thermionic emission current dominates below flat-band voltage [106].

4. Metal-Oxide-Semiconductor devices

4.1 Comparative study of MOS stack on oxygen and hydrogen terminated diamond

The Ideal MOS structure requires an electrostatic potential barrier which hinders the carrier transport from the semiconductor to the gate metal. Depending on the carrier transport mechanisms and on the specific characteristics of the device, it could also be possible to provide a single potential barrier with respect to the conduction or the valence band. This is the case of H-terminated diamond FETs where a single barrier for 2D hole gas is required [116]. Double and triple oxide stacks have also been

investigated on H-terminated diamond surfaces and their electrical properties (i.e. hysteresis, band offset, leakage) accurately reviewed in [117]. A schematic band alignment computed with the procedure and the parameters defined in [118, 119] has been plotted in figure 8 for both H-terminated(a) and O-terminated(b) diamond. As it can be noted, only few oxides (like SiO_2 , Al_2O_3) would allow for a dual barrier with the conduction and valence band of O-terminated diamond. Moreover, despite the fact that numerous diamond-oxide interfaces have been studied in the last few years [117, 120], the lack of native oxides for diamond has often resulted in highly defective interfaces which have negatively impacted on the carrier mobility at the interface.

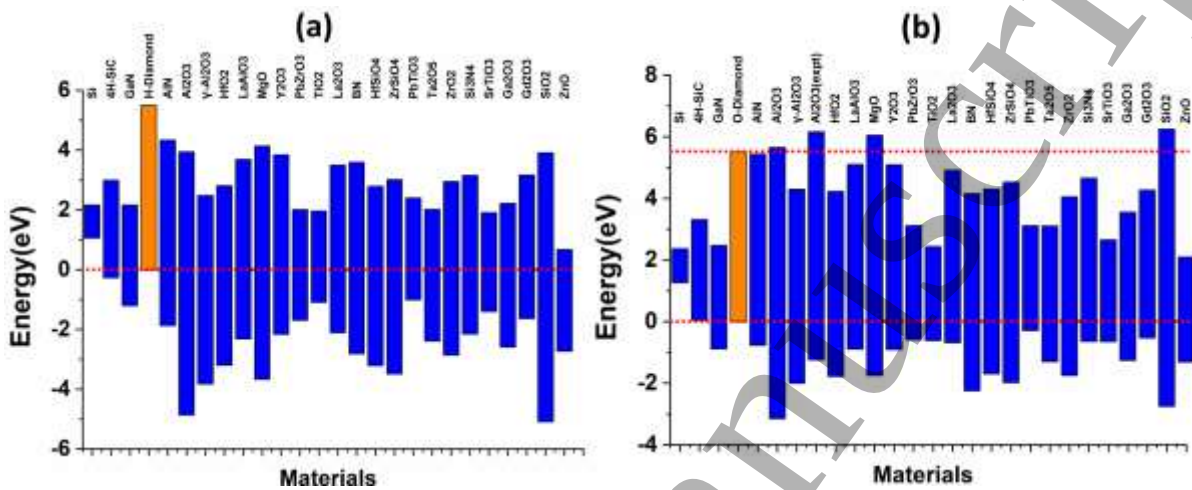


Figure 8, Band alignment of H-terminated (a) and O-terminated (b) diamond with several oxides. Calculation is based on the parameters and the procedure adopted by Robertson and Monch [118, 119]. The NEA for H-terminated diamond results in a negative conduction band offset (i.e. no barrier for electrons). Experimental Al_2O_3 /O-Terminated diamond alignment reported in [121] has been included in figure 8(b). The experimental bandgap of Al_2O_3 on diamond is smaller compared to the value reported by Robertson in [119].

4.1.2.1 Oxygen terminated diamond MOS regimes and reliability

Experimental results have shown that Al_2O_3 exhibits the best performance in controlling O-terminated diamond interfaces. The electrical properties and the band alignment of this stack have been reported in [121]. Leakage current mechanism occurring in Diamond/ Al_2O_3 /Al has also been investigated by Pham et al. [122]. Only recently, a few reliability studies have been reported for diamond MOS stacks. In [123], Loto et al. have observed a strong impact of the interface defects in the flat band voltage shift by means of time dependent bias stress. In addition, it has been demonstrated that the post process annealing improves the electrical performance of the MOS capacitor, with a clear accumulation regime observed even at relatively low negative bias and with a negligible gate leakage current value.

4.1.2.2 Deep depletion and inversion mode MOSFETs

Among the properties correlated with the wide value of the diamond' bandgap (5.47eV at RT), the small value of the intrinsic carrier concentration has a positive effect on reducing the thermally generated minority carriers for the creation of inversion regime formed in the MOSFET devices [124]. More specifically, if minority carriers are not provided by source and drain regions or by UV light exposure, a deep depletion regime can be obtained for a long and stable duration. The concept of temperature-time stable deep depletion effect observed and demonstrated for diamond devices is different from the dynamic effect described in other semiconductors like silicon [125-128]. In that case, the inversion layer is much more sensitive to time and temperature effects, making deep

depletion only a transient effect able to improve the dynamic BV. Experimental deep depletion diamond MOSFETs rely on the $\text{Al}_2\text{O}_3/(\text{Ti}/\text{Pt}/\text{Au})$ stack. High breakdown fields of 4MV/cm have been measured for a lateral normally-ON device schematically depicted in figure 9. However, the maximum current density observed in [129] is several orders of magnitude lower than the one reached for H-terminated FETs.

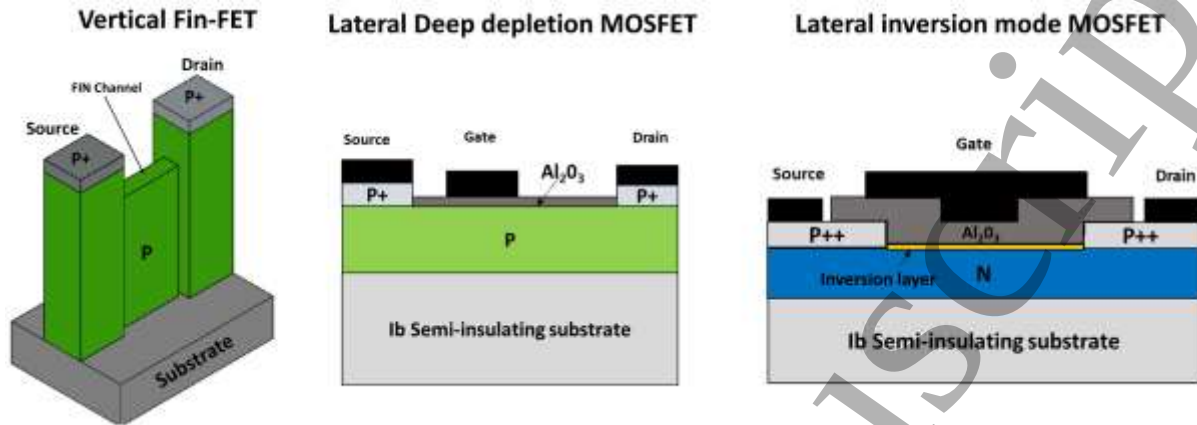


Figure 9, 3D Schematic of the diamond depletion mode MOSFET in vertical FinFet configuration and cross section of a lateral deep depletion and inversion mode MOSFET. The gate dielectric and metal surround the whole FIN channel (not shown in the picture).

Table 7. State-of-the-art parameters and key features for diamond depletion and inversion mode MOSFETs depicted in figure 9.

Device	Vertical Fin-FET	Lateral Deep depletion MOSFET	Lateral inversion mode MOSFET
Breakdown voltage	>16V at T=300K [130]	> 200V at T=300K [129, 131]	<50V at T=300K [73]
Current Density*	<p>< 1mA/mm at T=300K <10mA/mm at T=450K with VDS=-15V and VGS=-16V [130]</p> <p><0.05 mA/mm T=300K <2mA/mm at T=450K with VGS=-10V VDS=-1V</p>	<p>≈0.1mA/mm at T=300K with VDS=-15V and VGS=-16V [131]</p> <p><1e-3 mA/mm at T=300K ≈5e-3 mA/mm at T=450K with VGS=-10V VDS=-1V</p> <p>With selective growth of P+: ≈3mA/mm at T=523K with VGS=0V VDS=-1V</p>	<1mA/mm at T=300K With VGS=-5V VDS=-1V [73]
Notes	<p>Fin channel allows for normally-OFF operation.</p> <p>Breakdown voltage measurements are not reported. However, Gate and drain overlap limits the max BV.</p> <p>Max observed drain current is limited to 838nA for VGS=VDS=-16V at T=300K and 29μA at 450K [130].</p>	<p>Normally-ON.</p> <p>High temperature operation increases the current and reduce the threshold voltage.</p> <p>Scaling of the Ron_spec and BV is an issue.</p> <p>Field Plates are needed to improve the BV.</p> <p>Current density is limited by the incomplete ionization at RT.</p>	<p>Normally-OFF.</p> <p>Low interface mobility.</p> <p>High density of traps.</p> <p>Low breakdown voltage.</p> <p>First proof of concept.</p>

*Values reported for the Fin-FET are assuming the true width of the current transport path.

Recent reports have demonstrated the possibility of realizing deep depletion diamond Fin-FET with CVD boron doping on a $\langle 100 \rangle$ $3 \times 3 \text{ mm}^2$ HPHT undoped substrate. E-beam lithography and O_2 dry etching have been used to fabricate the Fin-FET structure depicted in figure 9. The low value of the boron concentration in the channel ($5 \times 10^{16} \text{ cm}^{-3}$) together with the 45nm of SiO_2 oxide and the small metal work function of Al ($\approx 4.08 \text{ eV}$) result in a depletion width of about 55nm. As the depletion region width is more than half the Fin channel width, the device exhibits normally-OFF behavior.

High quality Phosphorous doped n-type diamond body/ Al_2O_3 interface obtained by wet annealing has resulted in the first diamond Inversion type lateral MOSFET on a $\langle 111 \rangle$ HPHT substrate, as reported by Matsumoto et al. [73]. A maximum drain current density of 1.6mA/mm and channel field effect mobility of $8 \text{ cm}^2/\text{Vs}$ have been extracted from the experimental data at $\text{VGS} = -12 \text{ V}$ and $\text{VD} = -5 \text{ V}$ (table 7). This proof of concept for an inversion mode MOSFET resulted in a normally-OFF behavior, with a negative threshold voltage (Vth) of about -6.3V. This high Vth value for the inversion regime is a clear signature of high level of interface traps (with a density estimated to be above $6 \times 10^{12} \text{ cm}^{-2} \text{ eV}^{-1}$).

4.1.4. JFETs, MESFETs and Bipolar Transistors

FETs based on metal semiconductor junction (MESFET) or pn junction (JFET) are highly reliable for power electronics applications due to the absence of the gate oxide layer which tends to generate high density interface states and trapping/de-trapping mechanisms (figure 10). Umezawa et al. [132] have fabricated several diamond MESFETs, exploring different Schottky gate metals (Mo, Pt, Al) and observed a maximum current density of 1.2mA/mm at high temperature ($\text{T} = 600 \text{ K}$) with $\text{VGS} = 0 \text{ V}$ and $\text{VDS} = -20 \text{ V}$ due to the enhanced boron activation in the conduction region. High breakdown voltages above 2kV with a gate to drain distance of $50 \mu\text{m}$ [133] have been shown for diamond MESFETs which usually exhibit normally-ON characteristics with a high threshold voltage (Vth). Good scalability of the breakdown voltage with the gate drain distance has been proven for diamond MESFETs [132]. Diamond MESFETs have been also realized in reverse blocking (RB) configurations with a Schottky metal for the gate and the drain contacts [134].

On the other hand, improvements in the lateral growth of n-type diamond layer in the $\langle 111 \rangle$ direction have enabled the fabrication of high quality diamond pn+ junction with high rectification ratio and breakdown voltages close to 1kV [37, 39, 48, 135, 136]. These pn+ junctions have been used as the building block of diamond Junction FET fabricated by Hosino et al. [39]. Different channel width and doping levels have resulted in both normally-ON and normally-OFF devices demonstrated in unipolar and bipolar conduction mode. The BV measured at different junction temperatures shows a positive coefficient, according to the increase of the phonon scattering and consequent reduction of the avalanche multiplication coefficient [135]. Normally-OFF JFETs (with a Vth around -1.2V) have been manufactured by implementing a parallel reduction of the doping concentration and the channel width ($\approx 0.2 \mu\text{m}$) in order to pinch-off the channel at zero bias. Devices show a good rectification ratio but a much smaller current density due to the higher resistivity of the channel region [137]. Despite the current density increases at HT, a positive shift of the threshold voltage with the temperature, which has been also confirmed by TCAD simulations and experimental results [35, 138], may however result in normally-ON operations at elevated temperature. Improvements in terms of current densities (table 8) have been achieved with both normally-ON and normally-OFF JFET operating in bipolar mode, with the injection of minority carriers (electrons) in the p-type region. However, the bipolar conduction also increases the number of carriers in the channel and this would result in a slower turn OFF and a more complex gate driving technique.

The recent progress in the n-type doping technology has also allowed the fabrication of bipolar junction transistors (BJTs) [139-141]. Indeed, early fabrication processes have failed to demonstrate

the bipolar mode operation due to the high resistivity of the n-type base layer (around 10^{18}cm^{-3}) and the low diffusion length of minority carriers (holes) in the base region. The introduction of the n+ layer has enabled both the hopping conductivity and the reduction of the series resistance due to the ohmic contact of the base. However, scalability of these devices is highly limited due to the low diffusion length [140].

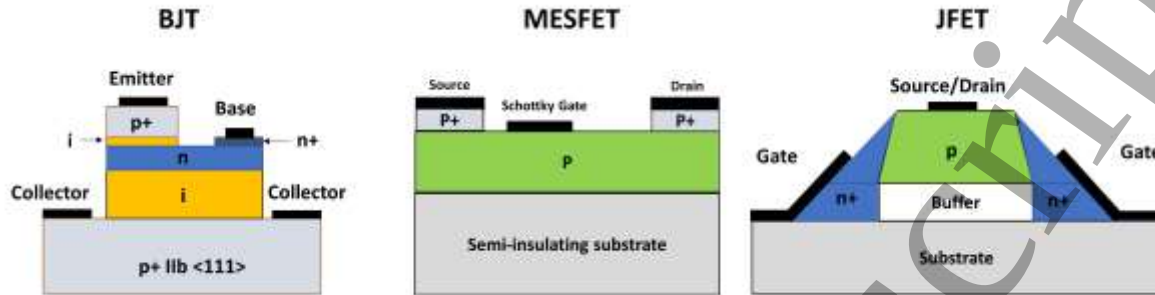


Figure 10, Schematic cross sections of a diamond BJT, MESFET and JFET.

Table 8. State-of-the-art parameters and key features for diamond BJTs, MESFETs and JFETs shown in figure 10.

Device	BJT	MESFET	JFET
Breakdown voltage	>100V at T=300K[140]	>2kV at T=300K[133] ≈3kV at T=300K for RB MESFET [134]	>600V at T=300K [135]
Current Density *	Not reported. Max current ~μA at $V_{EB}>6\text{V}$ [139].	≈2 mA/mm at T=500K with $V_{DS}=-20\text{V}$ $V_{GS}=0\text{V}$ [142] ≈0.14 mA/mm at T=300K with $V_{DS}=-20\text{V}$ $V_{GS}=0\text{V}$ [142] Max current ≈30mA at T>550K [6] ≈0.1mA/mm at $V_{DS}=-1\text{V}$ $V_{GS}=0\text{V}$ at T=600K [132, 142]	max current (≈2μA) at T=573K with $V_{DS}<-10\text{V}$ [138, 143] 600A/cm ² at T=500K 40A/cm ² at T=300K with $V_{DS}=-1\text{V}$ and $I_g=0.2\mu\text{A}/2\text{nA}$ (bipolar mode) [138, 143] ≈3.5kA/cm ² at T=500K with $V_{DS}=-20\text{V}$ and $I_g=1\mu\text{A}$ (bipolar mode) [143]
Notes	Need good doping control of both n-type and p-type layer. Lifetime control is needed for high current gain. Low BV.	Radiation hardness even at high junction temperature. Easy fabrication process (only requires p-type doping). Good scaling of the BV with the drift layer length.	Bipolar mode operations and normally-OFF demonstrated. Positive temperature coefficient of the breakdown voltage. Requires n+ doping. High temperature operations improve the R_{on_spec} but the temperature can seriously affect the operation mode (normally-OFF becomes normally-ON).

*The reported current density for diamond lateral JFET is normalized with the cross-sectional area.

4.1.5. Two-dimensional hole gas-based FETs

Two-dimensional hole gas (2DHG) formation near the hydrogen terminated diamond surfaces provides an innovative way to obtain an almost zero activation energy hole channel. This effect which was unveiled in the early 1990s [144-148], has been discovered to be useful for the fabrication of surface channel FETs [54]. On top of that, the maximum measured channel mobility typically around $100\text{cm}^2/\text{Vs}$ and the sheet hole density which oscillates between 10^{12}cm^{-2} up to 10^{14}cm^{-2} (with NO_2 adsorption [149, 150]) are promising electrical properties for the next generation of diamond power devices [151].

Atomic layer deposition (ALD) of Al_2O_3 have been proven to be a new way to uniformly induce the hole accumulation layer and improve the overall reliability and stability of the HFETs [149, 152]. Compared to the surface adsorbates, the insulating layer possess some unoccupied orbitals or fixed negative charges which are responsible for the formation of the 2DHG at the interface [153]. Lateral normally-ON HFET with a high breakdown voltage (over 1.5kV) and high temperature ($>725\text{K}$) stability have been reported in the literature [57, 154-159]. Lateral triple-gate HFETs which allow carrier to flow in both lateral and planar directions have illustrated higher current density and more promising downscaling scenarios compared to classic lateral HFETs [160].

Several solutions have also been implemented to avoid the formation of the 2DHG under the gate region and achieve the enhancement mode behavior. For example, Liu et al. [155, 161] deposited a double high-k layer oxide to avoid the formation of unoccupied levels and remove the 2DHG from the gate region while Kitayabashi et al. [11] obtained the normally-OFF operations with the partial oxidation of channel region with C-O bounds. Because the hole sheet created at the diamond interface is not based on piezo-polarization effects as in $\text{AlGaIn}/\text{GaIn}$ interfaces, it can be formed in non-planar structure (i.e. vertical trenches), as it has already been reported by Inaba et al. and Oi et al. [162, 163] (figure 11). Temperature dependence of the leakage current still remains a fundamental issue with HFETs due to the residual doping concentration in the bulk region and the lack of proper isolation. In-situ annealing performed prior the oxide deposition of the hydrogenated diamond surface at $\sim 675\text{K}$ was found to be crucial to enhance long-term doping stability of HFETs fabricated on MoO_3 and V_2O_5 , as reported in [164]. This evidence opens a promising route for the high temperature applications and for the possible future commercialization of diamond HFETs.

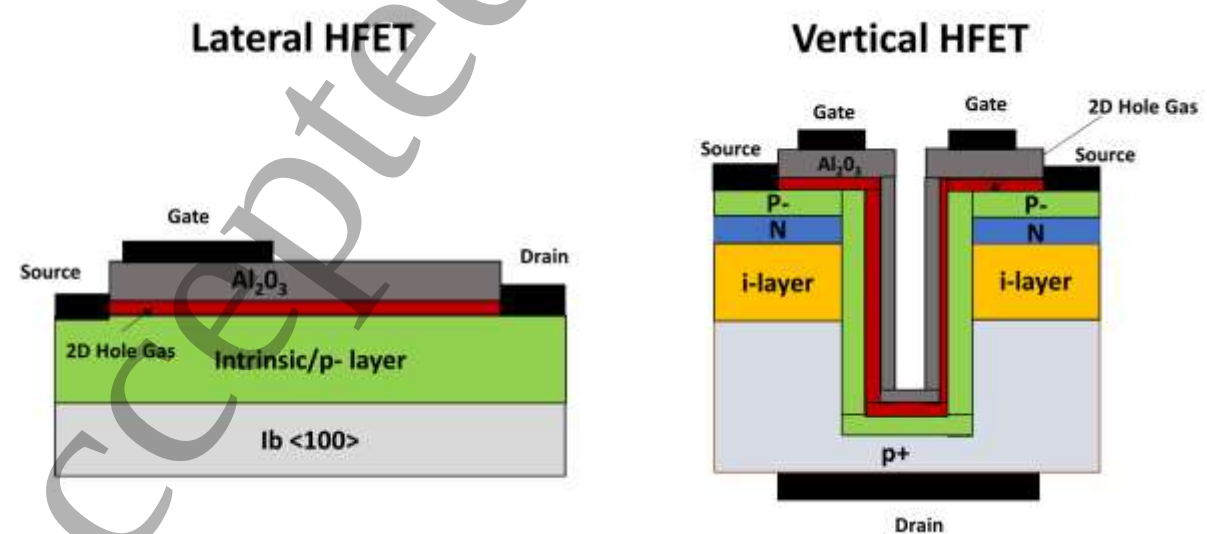


Figure 11, Schematic cross sections of diamond lateral and vertical HFET.

Table 9. State-of-the-art parameters and key features for diamond lateral and vertical HFET depicted in figure 11.

Device	Lateral HFET	Vertical HFET
Breakdown voltage	>2kV at T=300K [11]	≈350V at T=300K [162, 163]
Current Density	1.3A/mm at T=300K with VGS=-5V and VDS=-12V [165] 0.2A/mm at T=300K with VGS=-5V and VDS=-1V [165]	>0.2A/mm at T=300K and T=600K with VDS=-50V and VGS=-20V [162] <10mA/mm for VDS=-1V and VGS=-5V [162]
Notes	Lateral current flow limits the scalability of the ON state resistance. BV scalability is limited. Beneficial for RF applications [157, 166]. Normally-OFF has been demonstrated. Fabricated with both poly and mono crystalline diamond.	The beneficial vertical current spreading only starts from the p+ layer. BV is limited. Complex fabrication process which requires deep etching. n-type layer specifications are crucial to reduce vertical leakage current.

4.1.6 Vacuum Switches

Hydrogen terminated diamond interfaces are well known to exhibit a unique property renowned in the literature as NEA (Negative Electron Affinity), already discussed in paragraph 2.3.2. This feature is very attractive for the realization of electron emitters as, from a theoretical point of view, electrons excited from the valence band or injected from contacts into the conduction band could be efficiently emitted in vacuum from the surface without any increase of the device temperature [167-171].

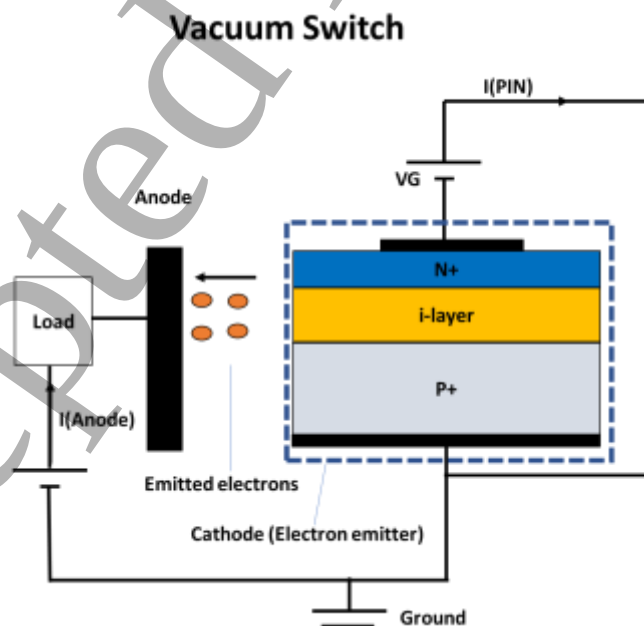


Figure 12, Schematic cross sections of diamond Vacuum switch and its control circuitry.

Table 10. State-of-the-art parameters and key features for diamond vacuum switch illustrated in figure 12.

Device	Vacuum Switch
Breakdown voltage	10kV at T=300K [168]
Current Density	4A/cm ² at VG<-20V at T=300K [170]
Notes	Unique device concept. Maximum current ~mA. Exploit the NEA of diamond. Efficiency and output capacitance (Coss) need to be improved. Gate drivers may result in non-conventional designs for power electronics.

Experiments have confirmed electron emission from diamond pn and PIN diodes with an efficiency oscillating around 2%. A schematic representation of the diamond vacuum switch fabricated in [168, 171] is illustrated in the figure 12.

4.2. Comparison of 2DHG and bulk transistors

A fair comparison between 2DHG based transistors and bulk doped ones (deep depletion or inversion MOSFET, MESFET, JFET) is very important for the optimization of the current diamond FETs' topologies for the next generation of power devices. Firstly, the Ron_spec vs BV dependence needs to be carefully analyzed for different operating temperatures. At RT, the high 2DHG concentration gives rise to a total resistivity for HFETs which is much lower than diamond bulk FETs. Indeed, while bulk diamond FETs tend to exhibit a much higher bulk mobility, their carrier density is seriously affected by the incomplete ionization effect. Besides, while the HT effects are almost negligible on the Ron_spec of 2DHG channels, their impact on bulk diamond FETs is much more significant due the enhanced activation of dopants (figure 3a and 4). Figure 13 shows the structure of the compared devices in the case of depletion mode MOSFET, with the consequent expression for the specific ON state resistance Ron_spec. Equations (7) and (8) shows the relationship between sheet resistance Rs, and hole mobility μ_{2DHG} and sheet concentration in 2DHG $N_{sheet2DHG}$ (eq. 7), and bulk hole mobility μ_{Drift} and concentration p_{Drift} and drift region thickness d_{Drift} . The values of μ_{2DHG} and $N_{sheet2DHG}$ are plotted in figure 3b. The values and dependences of μ_{Drift} are plotted in figure 3a and 4. For both devices, equation (9) shows the relationship between the specific ON state resistance Ron_spec, the sheet resistance Rs and the drift region length L_{Drift} .

2DHG lateral:	$R_S = \frac{1}{q \cdot \mu_{2DHG} \cdot N_{Sheet2DHG}} \quad [\Omega/\square]$	(7)
Bulk lateral: <i>Assuming a constant current density through the whole drift region thickness</i>	$R_S = \frac{1}{q \cdot \mu_{Drift} \cdot d_{Drift} \cdot p_{Drift}} = \frac{\rho}{d_{Drift}} \quad [\Omega/\square]$	(8)
Lateral devices:	$R_{on_spec} = R_S \cdot L_{Drift}^2$	(9)

Figure 14 shows the comparison of lateral bulk and lateral 2DHG devices, for 1kV and 3kV cases, at various temperatures. In these plots, only the drift region resistance is considered, incomplete ionization of boron and hole mobility dependence with doping and temperature is modeled [70], bulk and 2DHG devices have the same L_{Drift} value (NPT condition as predicted by [172]). For bulk devices, the relationship between the hole concentration p_{Drift} and the boron concentration N_{A0} was presented earlier in fig.2a). For 1kV, the parameters of the drift region are $L_{\text{Drift}} = 1.9\mu\text{m}$ and the boron concentration N_{A0} equal to $1.8 \times 10^{17}\text{cm}^{-3}$. For 3kV, the parameters of the drift region are $L_{\text{Drift}} = 10\mu\text{m}$ and $N_{A0} = 1.9 \times 10^{16}\text{cm}^{-3}$.

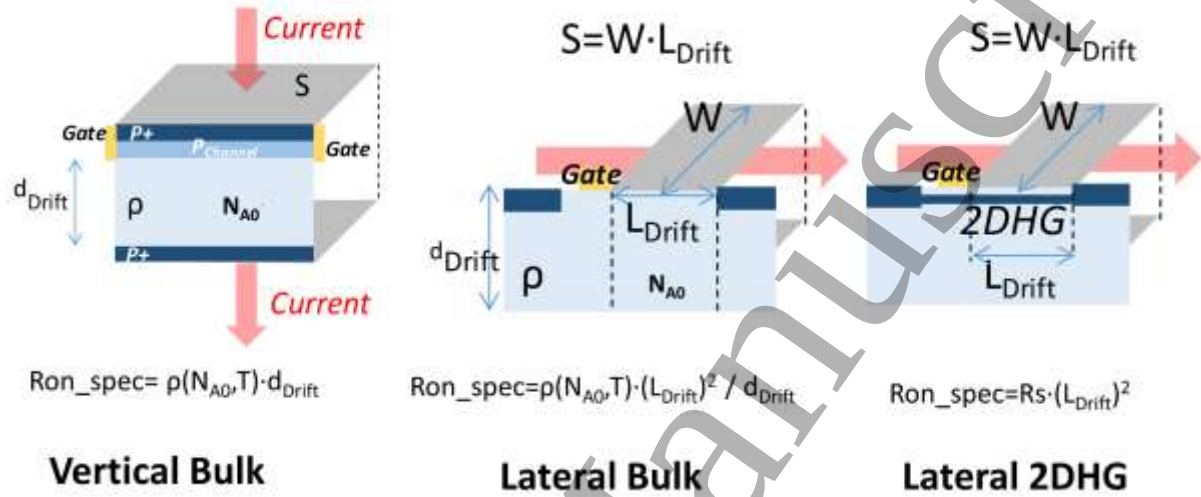


Figure 13, Surface definition and specific ON state resistance expression for vertical bulk, lateral bulk and lateral 2DHG based diamond devices. Only the drift region is considered, as the main source of total series resistance.

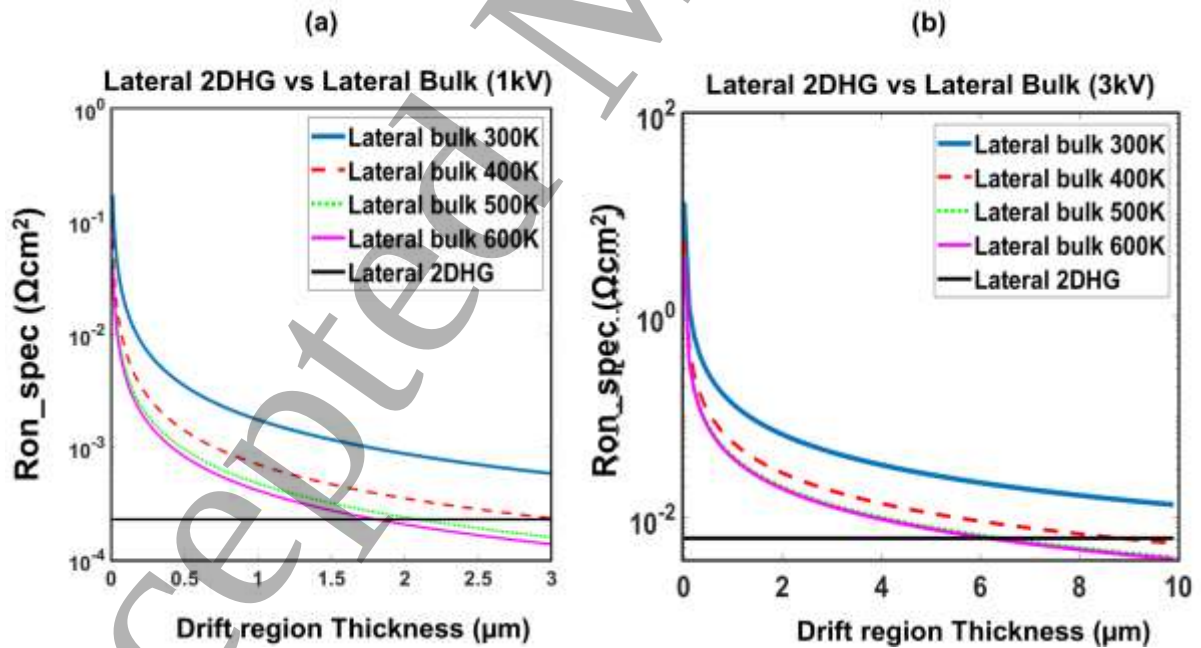


Figure 14, $R_{\text{on_spec}}$ vs drift region thickness of diamond lateral devices. Assumptions are $\mu = 100\text{cm}^2/\text{Vs}$ and constant hole sheet density of 10^{13}cm^{-2} for H-terminated diamond FET ($R_s = 10\text{k}\Omega/\square$) and mobility-carrier temperature dependent parameters for oxygen terminated diamond bulk FETs. a) 1kV, b) 3kV. As one can observe, the bulk region of a diamond FET needs to be $>1.7\mu\text{m}$ and $6\mu\text{m}$ for 1kV and 3kV BV, respectively, in order to obtain a reduction of the ON state resistance compared to HFETs.

Typically, lateral 2DHG will have a non-intentionally doped (NID) drift layer, which would result in a reduced peak electric field at the breakdown and a wider drift region, with a consequently higher R_{on_spec} (not considered here). The resistivity in lateral 2DHG is assumed to be independent with temperature, as evidenced by experimental reports in [173]. As one can note from figure 14, there is a minimum drift region thickness for which the lateral bulk devices will have a smaller R_{on_spec} than lateral 2DHG, depending on the BV and junction temperature. The R_{on_spec} for vertical devices is plotted at RT and 450K in figure 1a-b. This analysis must then be integrated in a wider investigation, considering additionally voltage threshold tuning, gate and drain leakages, stability and reliability.

4.3 Leakage current in diamond devices

The increase of the junction temperature, which can enhance the conductivity in Diamond layers affected by the incomplete ionization of the dopants, poses some important questions about the specific leakage current mechanisms occurring in actual devices. Indeed, such leakage currents, which are typically higher for increased junction temperature, can lead the devices into premature breakdown phenomena. While the most significant performance degradation has been observed in all Diamond devices (e.g. SBDs) with a substantial number of dislocations and non-epitaxial crystallites [174], different leakage mechanisms occur in each specific device.

Regarding Diamond Schottky diodes, thermionic field emission leakage current with barrier lowering has shown good agreement with experimental results even at elevated junction temperature [21, 96, 175]. On the other hand, analysis on Diamond PIN diodes reverse characteristics between $323K < T < 423K$ have suggested that Poole-Frenkel emission dominates at high electric field while hopping conduction is the dominant mechanism at low electric field [102, 176].

On the other hand, the leakage current of MOS based devices typically depends on the number and nature of interfacial traps at the diamond/oxide interface [121]. Leakage current mechanism occurring in oxygen terminated Diamond/ Al_2O_3 /Al has been investigated by Pham et al. [122]. Authors suggested a 4-step mechanism responsible for the negative bias leakage current which originates from the hole carriers accumulated at the interface and it involves trap-to-trap tunneling in the oxide and charge transfer with the interface states. Thermal annealing of the gate oxide has proven to reduce leakage current values in diamond oxygen terminate MOS-based device. With such a thermal process, the gate leakage current value was reduced upto $< nA/mm$ ($100nA/mm$) with $V_{GS}=45V$ and $V_{DS}=-175V$ at RT ($T=500K$). To date, the same analysis has not been carried out yet for hydrogen terminated devices.

Lateral FET devices such as MESFETs and JFETs also suffer from buffer-related leakage current which usually increase at high temperature due to the enhanced activation. Such leakage mechanism is also present in lateral HFET in which also the unintentionally doped substrate can play a key-role at elevate temperature and increase the overall leakage in the device. To date, only few studies have been dedicated to leakage current in diamond FETs. Lack of proper isolation between individual device can also contribute to leakage at high voltage values. Reactive Ion Etching techniques can indeed enhance sidewall leakage and create shorts and techniques such as partial mesa etch are usually preferred.

4.4 Diamond devices with field relief designs

Without field relief designs, the typical vertical peak electric field at breakdown in diamond devices is limited to 1 to 2.5 MV/cm [6, 46, 91, 112]. Therefore, the use of field relief structures is required in

order to improve the breakdown voltage capability and also to suppress the detrimental effects of the device termination on the field-enhanced leakage current mechanisms.

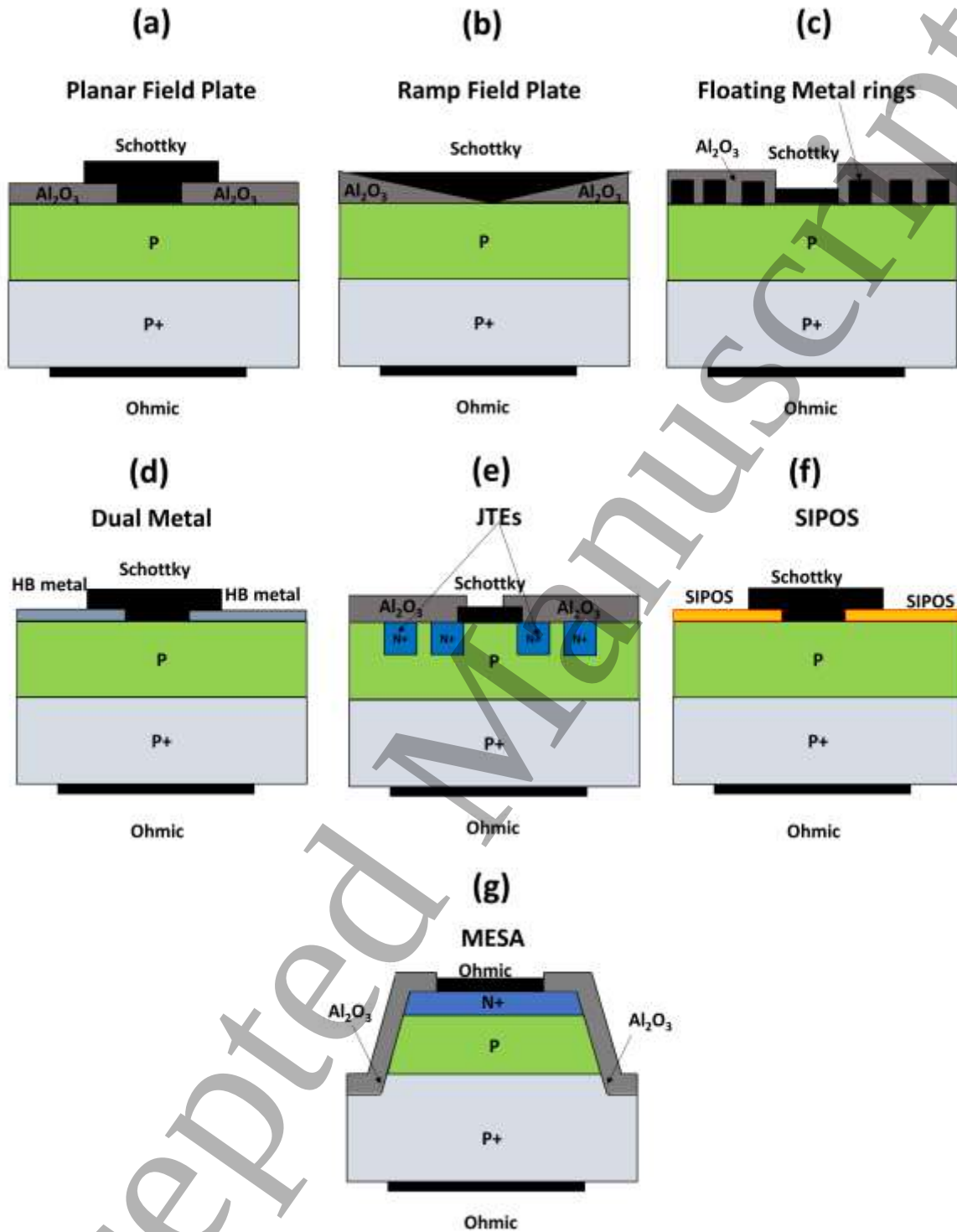


Figure 15, Different topologies for diamond terminations applied to the case of diodes. (a) Planar field plate, (b) Ramp field plate, (c) Floating metal rings, (d) Dual metal, (e) JTEs, (f) SIPOS, (g) MESA. Al₂O₃ has been used as oxide only as an example.

For unipolar mode diamond devices like SBDs, field plate structures (figure 15(a)) have been more often adopted in the literature [89, 90, 92, 177, 178]. Theoretical optimization for a single layer FP structure has been carried out by Ikeda et al. [91], showing that for a BV reached at a maximum

leakage current density of 10^{-4} A/cm^2 , an optimum oxide thickness can be obtained for Al_2O_3 ($\sim 1.5 \mu\text{m}$) and SiO_2 ($\sim 0.9 \mu\text{m}$). Experimental results on vertical diamond SBDs have illustrated both a reduction of the leakage current and an improvement of the BV by using $0.2 \mu\text{m}$ Al_2O_3 on top of a $10 \mu\text{m}$ p-type boron doped layer [46].

Ramp field plate oxides have been suggested to be one of the most effective ways to reduce the peak electric field in SBDs (figure 15(b)). Calibration of ramp FP termination by means of TCAD simulation has been performed by Brezeanu et al. [178] and shown an almost ideal BV (92% of efficiency).

Diamond Schottky diodes with floating metal rings (figure 15(c)) have also been manufactured by Driche et al. [179] and their efficient reduction of the electrostatic potential crowding has been confirmed by EBIC measurement. The spacing between the different rings and the number of rings influence the peak electric field and the shape of the lateral depletion. Due to the high field gradients in diamond, the reduced spacing of such rings induces a high stress on fabrication and lithography.

In diamond SBDs, the increase of the leakage current based on thermionic field emission effects [174, 180] could be efficiently tackled with a double metal termination (high barrier (HB) and low barrier (LB)), as shown in figure 15(d). By ensuring a low barrier in the central area of the structure while increasing it in the periphery of the structure (with metals like Au or Pt), it is possible to contemporarily suppress the value of the OFF state current and avoid any increase of the threshold voltage.

The lack of an efficient n-type doping and the issues arising from the ion implantation have Junction termination extensions (JTEs) less effective in diamond (figure 15(e)). Kubovic et al. [181] did not observe any improvement after the 10nm of n+ type nitrogen doped layer while Huang et al. [182] tried to obtain the same effect through H+ ion implantation to increase the resistivity, reporting a breakdown voltage of about 3.7kV for a diamond SBD.

Semi-insulating polycrystalline silicon (SIPOS) terminations provide a more uniform distribution of the field at the expense of an increased surface ohmic leakage (figure 15(f)). This kind of termination technique has been experimentally demonstrated for diamond SBDs and MESFETs [183]. MESA etching termination technique could be also adopted for diamond pn junctions, as already suggested in [110, 184]. However, one has to note that the optimal drift region thickness in diamond is larger than $10 \mu\text{m}$ for breakdown voltages above 3kV [172] (theoretical) and etching thick diamond is a difficult process as the whole drift region must be etched. There is also the possibility of sidewall leakage induced by defects during etch, usually performed by Deep reactive-ion etching (DRIE). These considerations currently limit the possibility of MESA termination in the context of high voltage diamond devices.

4.5 Packaging, thermal management and reliability

Due to the novel nature of diamond devices, no dedicated packaging technique has been developed yet. A suggested package solution for efficient thermal dissipation and its equivalent Spice DC thermal network for a diamond power semiconductor has been illustrated in figure 16. In the literature, only few diamond devices have been packaged and tested in power circuits. MESFETs in [89] have been packaged on a typical metal-ceramic package where the device was bonded with Au and molded with a resin. In [9], vertical type Schottky diodes have been packaged with a silicone-based resin which has been then hermetically sealed by the stainless-steel cover.

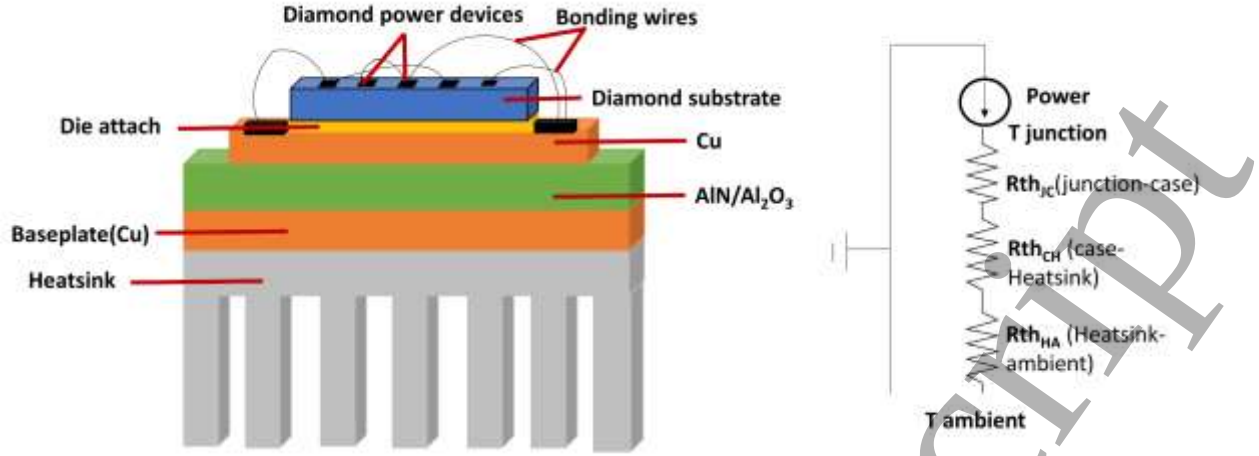


Figure 16, Schematic example of a package for a diamond device and its Spice DC thermal equivalent circuit.

As introduced hereinbefore, due to incomplete ionization, the R_{on_spec} of diamond bulk devices has a negative temperature coefficient (NTC). Furthermore, as demonstrated experimentally in [10], the switching losses are not affected much by increased temperatures. As a consequence, the total losses of diamond bulk devices have a NTC up to a high temperature where the losses are minimized.

This important NTC modifies the design of the heatsink with diamond devices, where self-heating can be used to increase the junction temperature and to reduce losses at the same time. Consequently, the $R_{th_{CA}}$ can be largely increased with diamond devices, leading at the same time to lower power losses and smaller and lighter heatsink.

Figure 17 shows the comparison for two diodes having the same breakdown voltage (1.2kV) and current rating (10A), based either on Diamond or 4H-SiC (commercially available SiC device: CPW4-1200-S010B from Wolfspeed). The device area for Diamond and 4H-SiC are respectively 0.37mm^2 and 3.92mm^2 [185]. In all thermal analyses, the temperature within each die is supposedly uniform and equal to the related junction temperature. The ambient temperature has been set to 300K.

The current flowing through the devices is 10A, with a duty cycle of 50%. Consequently, the relationship between the required total case to ambient thermal resistance $R_{th_{CA}}$, the junction temperature T_j and the losses dissipated by the devices are defined by equation (10). Here, only conduction losses are taken into account while neglecting the junction barrier height (only conduction losses due to the series resistance).

The diamond device is supposedly vertical, with a doping level of $1.2 \cdot 10^{17}\text{cm}^{-3}$ and a drift region thickness of $2.5\mu\text{m}$ as proposed in [172] under the NPT condition. The dependence of R_{on} with temperature has been taken from incomplete ionization and doping and temperature dependence hole mobility for diamond (figure 3a and 4), whereas taken directly from the datasheet for SiC.

$$R_{th_{CA}}(T_j) = \frac{T_j [K] - T_a [K]}{R_{on}(T_j) \cdot I^2 \cdot \delta} = \frac{T_j [K] - 300 [K]}{R_{on}(T_j) \cdot 50 [A^2]} \quad (10)$$

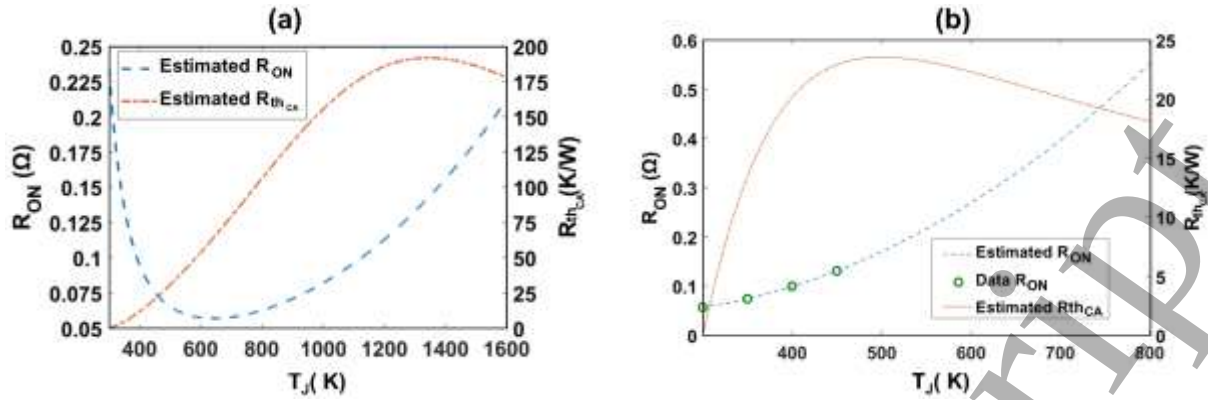


Figure 17, ON state resistance and required Case-Ambient Thermal resistance (R_{thCA}) for 1.2kV 10A power diodes, as a function of the junction temperature (T_j). A) Diamond, b) 4H-SiC.

As with other power devices with NTC coefficients, the thermal stability and current focusing possibility are serious issues. As an example, the parallelization of such devices can be challenging, especially when the diamond dies are poorly thermally coupled. Figure 18 introduces an example where two diamond Schottky diodes are parallelized, but have a 10% dispersion in R_{on_spec} , which could be due to process dispersion. In this analysis, the 10% dispersion in R_{on_spec} is modelled with different diode active areas and similar $J(V)$ characteristics. The initial R_{on_spec} and its dependence with temperature is taken from an actual diamond device [94], exhibiting a similar behaviour as figure 17a. The load current is 10A with a duty cycle of 50%.

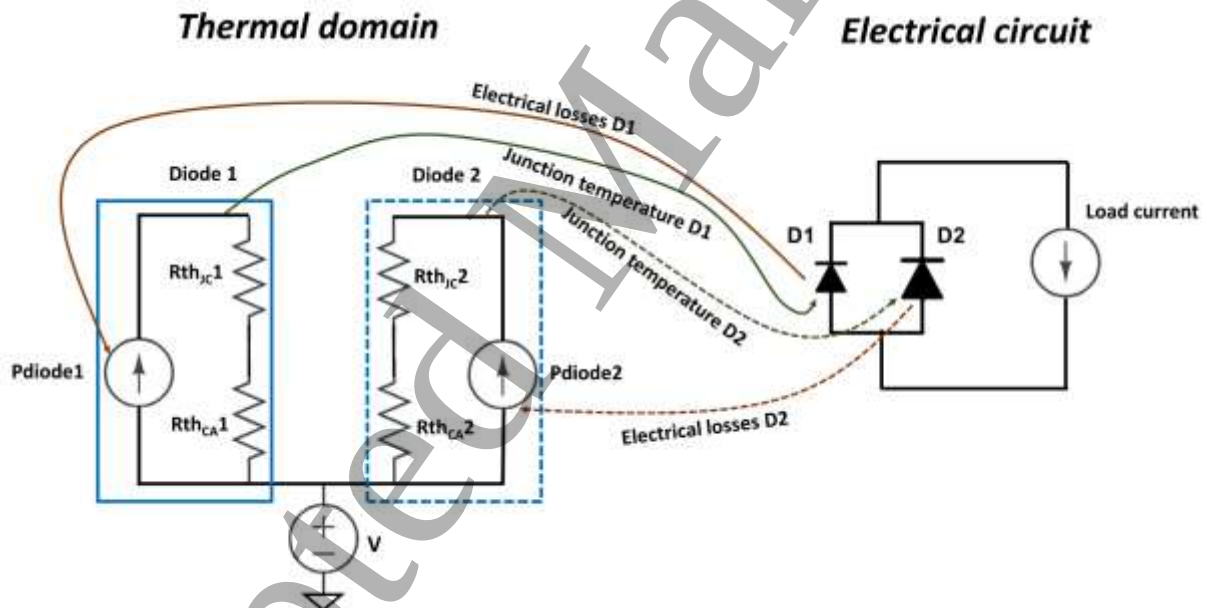


Figure 18, Electro-thermal coupling for two diamond diodes parallelized. D1 and D2 have a different size (D2 is 10% bigger than D1) as schematically depicted in the electrical circuit. Symbols for the equivalent thermal circuit are described in figure 19.

As a consequence of the different R_{on} (10%), most of the current is running through one diode (diode 2, 90% of the load current) and the other is conducting only a small part of the total current (only 10%). There is a clear current focusing effect due to the separate thermal heatsink and the small 10% dispersion in R_{on} of both diodes. Increasing the R_{thCA} value will lead to higher junction temperatures of each parallel diode and for the same operating point. However, the total current is more evenly shared between the two diodes, after an electro-thermal transient (Figure 19). Using larger R_{thCA} values is however not recommended to mitigate the characteristics dispersions between diodes, as it would lead to higher junction temperatures and limited surge current capabilities.

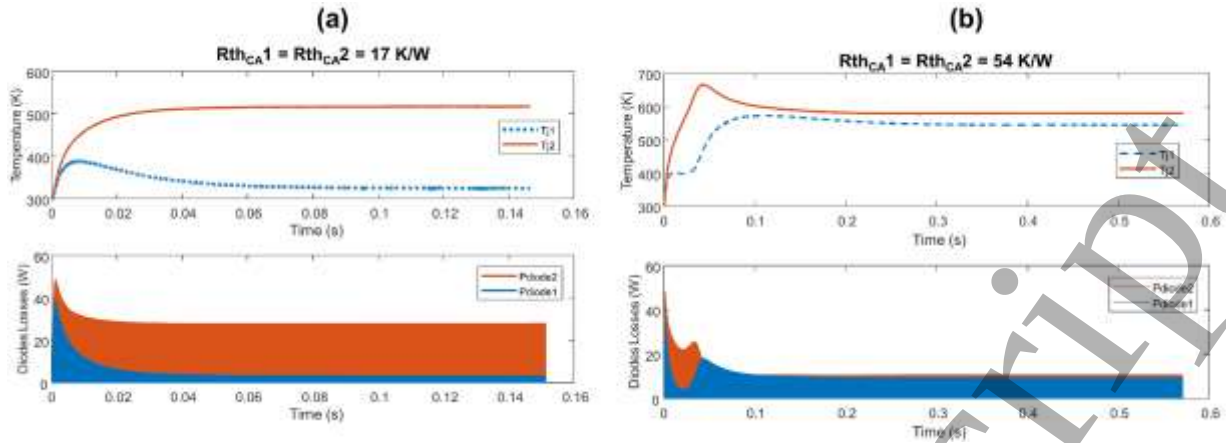


Figure 19, Transient simulation of temperature distribution and power loss for two diamond diodes in parallel, with a separate heatsink. (a) $R_{th_{CA}}=17\text{K/W}$ and (b) $R_{th_{CA}}=54\text{K/W}$.

The consequences of this simple example are that diamond devices must be thermally coupled in the best possible way and that specific thermal simulations must be investigated to optimize the paralleling of diamond devices and for diamond power modules. On the same diamond die, one can expect that active cells paralleling will not be an issue due to the highest thermal conductivity of diamond, albeit with further investigations still required.

Reliability is one of the main concerns in diamond devices. Indeed, as the future generation of diamond power circuits is supposed to be working in extreme temperature conditions and for high frequency and voltage at the same time, the requirements on the overall system stability are even more strict than the one for silicon. Time dependent dielectric breakdown (TDDB) needs to be properly addressed (as in GaN) as the increase of the electric field in the structure due to the high voltage ratings may lead to a time dependent failure of the protective layers. High frequency performance with fast dV/dt and dI/dt could be limited by stray inductances and capacitances with possible enhanced oscillations which may result in malfunctions or delay in the turn ON/OFF. Moreover, appropriate Gate driving techniques and the reduction of parasitics also need to be considered in the design of diamond devices.

5. Benchmark of diamond power devices

5.1 System level benefits and challenges

P-type transistors and diodes are the most promising diamond devices for future commercialization. In particular, the absence of high-performance p-type FETs in the existing power electronics market could open a specific opportunity for diamond. As an example, the co-package integration of diamond FETs with other FETs based on GaN, $\beta\text{-Ga}_2\text{O}_3$, AlN and 4H-SiC could represent a powerful solution for providing a smart IC avoiding external gate drives and thus reducing the parasitics inductances of external interconnections [153, 186].

However, there exists some obstacles to overcome for diamond before this material can meet commercial expectations. Despite the fact that diamond devices would have lower total losses than other semiconductors, the power density losses are increased. Consequently, there is a higher stress on thermal spreader, accentuated by the higher junction temperatures of diamond devices. The system level benefits and challenges of diamond devices can be summarized as follow:

Benefits

- Reduced total semiconductor losses as a consequence of lower ON state losses.
- Increased switching frequency due to smaller active areas and very fast switching with the consequent significant reduction in size and weight of passive elements used in filters.
- Higher junction temperatures, leading to smaller and lighter heatsinks or moving from liquid cooling to forced air or even natural convection.

Challenges

- Higher power loss density, requiring efficient thermal spreaders and thermal interfaces (i.e. complex thermal management).
- Limited maximum diamond device area.
- Efficient device parallelization and system turn-on.
- Reliability and reproducible performance.

5.2 Power converters with diamond devices

Diamond devices are usually small in size and therefore can conduct only low currents. From this perspective, parallelization becomes an essential technique for increasing the current flowing through diamond devices. Examples of diamond diode parallelization in a buck DC/DC converter have already been studied in [10] and also partially addressed in paragraph 4.5. In [10], the diamond pseudo-vertical Schottky diodes were connected to a common anode and had isolated cathodes. A high side commercially available Si MOSFET was implemented in the experimental setup for the double pulse test in order to match the requirement of current/voltage of the diodes under test. The presence of the Si MOSFET limited the maximum switching speed of the system together with the parasitics (i.e. capacitances and stray inductances). However, it is critical to have similar output capacitance between the high side device (silicon transistor) and the low side device (diamond Schottky diodes in parallel). Coupling parallel diamond devices with high breakdown voltage ($>100\text{V}$ or $>1\text{kV}$) and a very low current capability ($<1\text{mA}$) on the high side with other power devices made of silicon, SiC or GaN on the low side while maintaining similar output capacitance is very challenging. In [60], a high switching speed was observed (tens of V/ns) with reduced oscillations mainly due to the low value of the switching current and significantly high ON state resistances. Such a parallelization of diamond devices was also analysed for an interleaved configuration which eased the increase of switching frequency with benefits related to the output filter design and control bandwidth (figure 20). The interference between the diamond diodes integrated on the same chip highlighted the importance of device isolation.

The interleaved setup may represent a promising configuration for the next generation of diamond converters with the on-chip integration of parallel p-type FETs on the same substrate. Besides, asynchronous DC/DC buck converter could benefit with the addition of a diamond p-channel FET due to the simplified gate driving technique (for the high side switch). However, unbalances between the different devices may impact negatively on the overall speed and current of the final converter, as already pointed out for diamond SBDs (paragraph 4.4).

Bridge converters with an integrated diamond solution would ideally require n-type FETs to simplify the gate driving technique. Nonetheless, a smart on-chip integration of a gate driver for the low side p-type diamond FETs could partially solve the issue arising from the absence of n-type diamond FETs. Finally, isolated p-type diamond converters made with only p-type FETs and diodes could offer a different solution to tackle the gate driving' issues of bridge configurations and other converters' typologies.

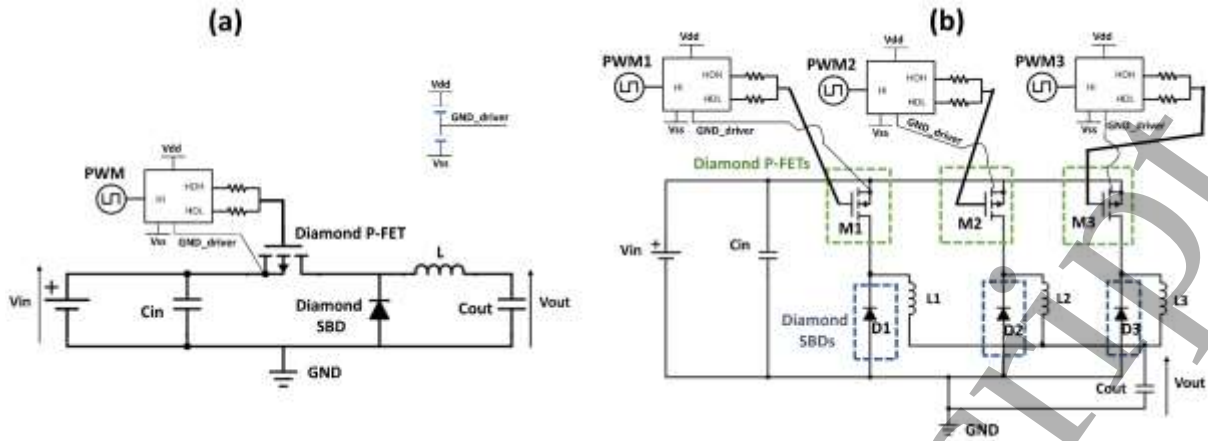


Figure 20, Schematic of (a) DC/DC buck converter and (b) interleaved converted with diamond p-type FETs and Schottky diodes. Depending on the switch, \$V_{ss}\$ can be positive and \$V_{dd}\$ can be negative. 3 separate gate drivers have been represented for the interleaved converter in figure (b). Alternatively, a single gate driver with separate inputs and outputs and one GND driver can be implemented.

5.3 Unipolar vs Bipolar diamond diodes

With many factors impacting on the electro-thermal performance and different physical mechanisms involved in the electron-hole current transport, the optimal choice between unipolar and bipolar devices for power electronics' applications needs be carefully carried out [187, 188]. One of the most accurate approaches to follow is the one described by Morissette et al. [187] for SiC diodes. This optimization is principally based on maximizing the available current density at a fixed breakdown voltage (BV) and switching frequency (f). If one assumes that the package and the cooling system cost between the two devices can be assumed identical, the only significant difference in cost is associated to the die. Therefore, higher current densities will allow for a reduced die area and lower cost for that specific device.

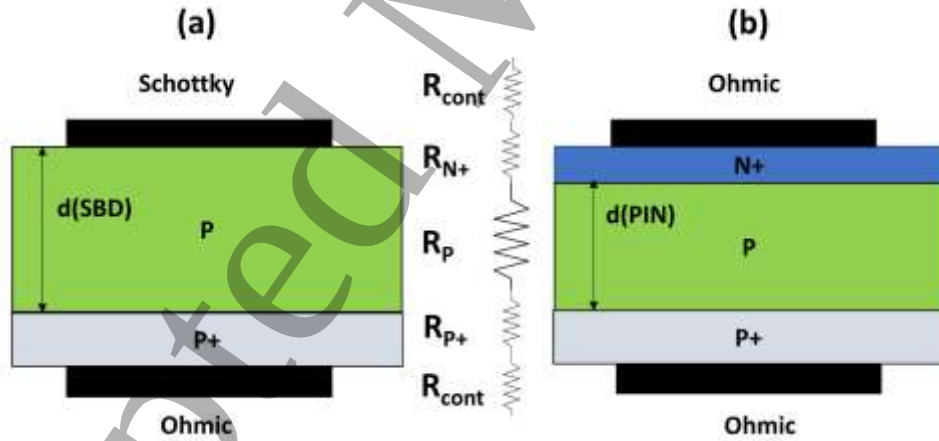


Figure 21, (a) Schottky diode and (b) PIN diode considered in the analysis. The resistance of the contact (\$R_{cont}\$), N+ and P+ type layers is neglected in this study (\$R_{N+}, R_{P+}\$). The doping of the N+ and P+ region is assumed to be \$10^{20} \text{ cm}^{-3}\$ (i.e. the incomplete ionization of these layers can be neglected).

In the analysis presented in this paragraph, a simple inductive load switching circuit is considered and the energy dissipated by the main switch (a FET such as a MOSFET/IGBT) is assumed to be directly based on the charge stored in the diode.

Under these assumptions, the static (\$P_{static}\$) and dynamic (\$P_{dynamic}\$) power density components for a power diode can be written as shown in equation (11-12) [74]:

$$P_{static} = J_F V_F \delta + J_R V_R (1 - \delta) \quad (11)$$

$$P_{dynamic} = f(E_{on} + E_{off}) \quad (12)$$

Where J_F is the current density in the ON state, V_F is the forward voltage drop, δ is the duty cycle (assumed equal to 0.5 in this study), J_R is the reverse current density, V_R is the reverse voltage (assumed equal to the BV in this simplified analysis), f is the switching frequency, and E_{on} and E_{off} are the energies loss densities by the diode during the turn ON and turn OFF transient of the diode. In addition, V_F can be expressed as the sum of the built-in voltage (V_{bi}) and the specific ON state resistance of the diode (R_p) multiplied by the forward current density for the PIN diode. In the formula (13), the built-in voltage (V_{bi}) is function of the bandgap(E_G), the impurity concentration of the P, P+ and N+ type layers, the density of states in the valence and conduction band and the operating temperature [34]. A different (but similar) expression holds for SBDs, as shown in (14)[34, 96]. In (14), η is the ideality factor of the diode (here assumed equal to 1), V_{bn} is the barrier height and A^* the Richardson constant (assumed equal to 88A/cm²K² from the matching with experimental diamond SBDs[96, 97]).

$$V_{F(PIN)} = J_F R_p(PIN) + V_{bi}(T, N_{A0}, N_{D0}) \quad (13)$$

$$V_{F(SBD)} = J_F R_p(SBD) + \eta V_{bn}(T, N_{A0}, N_{D0}) + \frac{\eta k T}{q} \ln\left(\frac{J_F}{A^* T^2}\right) = J_F R_p(SBD) + V_{bi}(SBD) \quad (14)$$

Regarding the specific ON state resistance of the p region for the Schottky Barrier diode (SBD), the absence of minority carriers results in the expression (15) - where p is the active carrier concentration calculated by means of the incomplete ionization model and $d(SBD)$ is the optimal punch-through thickness as in [172].

$$R_p(SBD) = \frac{d(SBD)}{q p u_p} \quad (15)$$

Regarding the bipolar PIN diode, the specific ON state resistance can be expressed as in (16):

$$R_p(PIN) = \frac{d(PIN)}{q p u_p + \frac{(u_p + u_n) J_F \tau}{d(PIN)}} \quad (16)$$

where $d(PIN)$ is the optimum drift layer thickness and τ is the ambipolar lifetime.

Unlike in the SBD, the best set of coefficients ($d(PIN)$, N_{A0}) cannot be obtained with a closed form optimization and an iterative technique by means of TCAD simulations is therefore needed. For the purpose of this study, $d(PIN)$ has been calculated by solving the ionization integral with the coefficients from [189, 190] and extracting the minimum thickness which gives a specific BV for a fixed dopant concentration (N_{A0}) of $5 \times 10^{14} \text{ cm}^{-3}$. This choice of the doping concentration has been carried out to allow a good level of conductivity modulation in the p-type layer.

With the previous assumptions in mind, the charge density stored in the PIN diode (Q_s) can be expressed as the product $J_F \tau$ and the energy density associated with the reverse recovery of the PIN diode during the turn OFF can be expressed by (17):

$$E_{off}(PIN) = \tau J_F V_R \quad (17)$$

For the purpose of this study, E_{on} , which is the energy dissipated by the diode during its turn ON, has been neglected for both the Schottky and the PIN diode. Furthermore, the E_{off} component of the diamond SBD has been neglected as no significant stored charge needs to be removed from the p-type layer, which is only composed of majority carriers (holes in this specific example). Nonetheless, high switching frequencies (>100kHz) could have an impact on the dynamic power dissipations with

the flow of the displacement current in the diode. This component (P_{disp}) has been taken into account for both devices, as shown in equation (18) where ϵ is the dielectric permittivity of diamond which has been assumed frequency independent.

$$P_{disp} = \frac{f}{3} \sqrt{\frac{\epsilon q N_{A0}}{2}} (V_F + V_R)^{\frac{3}{2}} \quad (18)$$

Bearing in mind all the previous assumptions, the total power dissipation (static and dynamic) for both devices can be expressed as in (19) and (20):

$$P_{(SBD)} = (J_F V_{bi}(SBD) + J_F^2 R_p(SBD)) \delta + J_R V_R (1 - \delta) + P_{disp}(SBD) \quad (19)$$

$$P_{(PIN)} = (J_F V_{bi}(PIN) + J_F^2 R_p(PIN)) \delta + J_R V_R (1 - \delta) + f \tau J_F V_R + P_{disp}(PIN) \quad (20)$$

For the purpose of this study, the component J_R has been neglected in both formulas.

In detail, for the PIN diode, J_R is mainly due to thermal generation-recombination process and can be ignored if one assumes a good quality of the material (low leakage current due to dislocations and defects). Regarding the SBD, such a component of the leakage current needs to be carefully considered as it can be significant for high electric field due to the thermionic field emission process. For the purpose of this study, the maximum level of the leakage current for SBD has been fixed at $1 \mu A/cm^2$ and the optimal V_{bn} has been extracted using the procedure illustrated in [96] at different operating temperatures. The optimal V_{bn} allows to minimize the ON state voltage drop and at the same time to maintain the desired leakage current density value for a specific reverse voltage.

Moreover, the ambipolar lifetime value τ has been optimized in order to minimize the power density expressed in (18) by setting the first derivative of $P_{(PIN)}$ equal to zero (figure 22). The optimal value of the lifetime (τ_{opt}) emerges from the trade-off between the static power dissipation which is reduced for high values of lifetime thanks to the conductivity modulation effect and the increased dynamic power dissipation which increases for larger stored charge (Q_s).

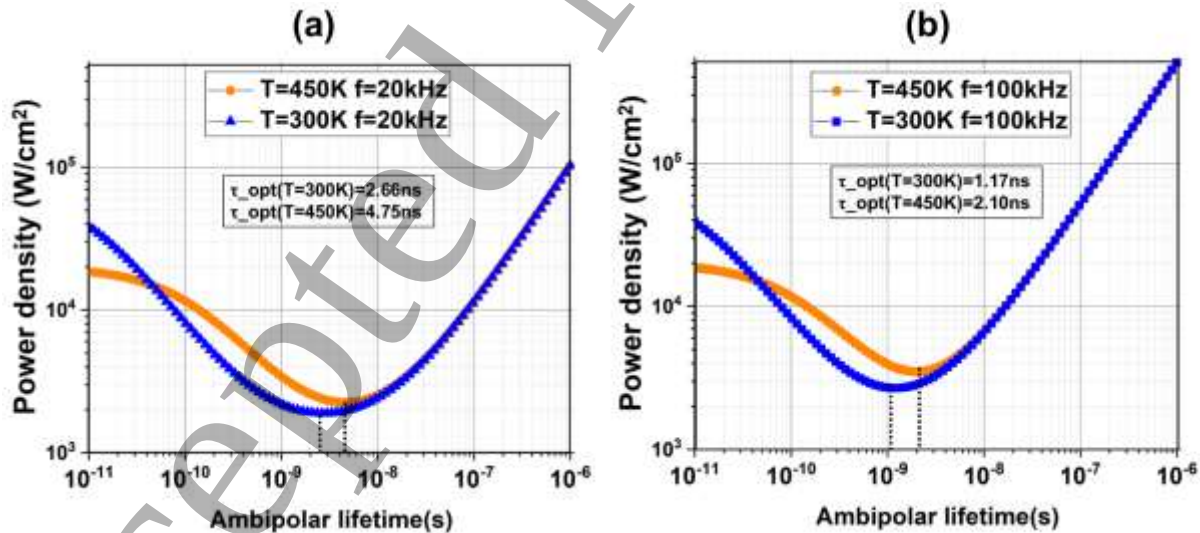


Figure 22, Power density vs ambipolar lifetime for PIN diode plotted with formula (18). For the plot it has been assumed a $BV=10kV$, constant J_F of $500 A/cm^2$ and (a) $f=20kHz$ and (b) $f=100kHz$.

Once the set of optimal parameters has been extracted, the procedure can be concluded by fixing a value for the maximum power density and by maximizing the J_F for each device by using formulas (17) and (18). The value of the maximum allowable power density typically depends upon the package capability (especially on the thermal spreader design) and it oscillates between $50-300W/cm^2$ for

commercial devices. In this paragraph, this value has been chosen to be $500\text{W}/\text{cm}^2$, a value which is justified by the increased thermal capability of diamond devices.

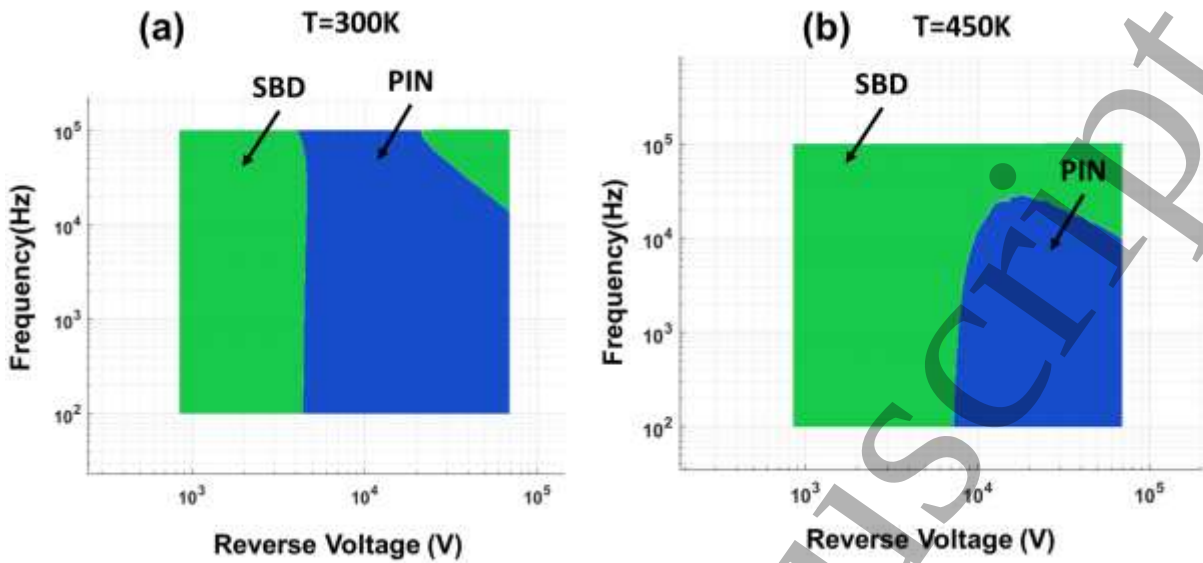


Figure 23, frequency vs reverse voltage domain for two different operating junction temperatures ($T=300\text{K}$ (a)/ 450K (b)). The areas in green are the ones where the SBD is a better choice than the PIN diode due to the higher current density.

At RT and for low breakdown voltage ($<4\text{kV}$), the current density for SBDs is higher than PIN diode and the SBD is the preferred device for the whole range of frequencies (figure 23). As the BV increases, the “optimized” conductivity modulation occurring in the PIN diode allows for reduced power losses. For ultra-high BV ($>20\text{kV}$) and for switching frequencies $>10\text{kHz}$, the SBD becomes again superior to the PIN diode. At higher junction temperature ($T=450\text{K}$) the trade-off between the two devices is modified due to the dependence of the leakage current upon the temperature, the carrier activation, the carrier mobility, density of states etc. which modify the set of optimal parameters for the analysis. The voltage vs frequency area in which the SBD displays higher current density compared to the SBD is widened and the PIN diode becomes a better choice only for reverse voltage $>7\text{kV}$ and $f<10\text{kHz}$.

5.4 Benchmarking of Diamond devices against 4H-SiC and GaN

In this paragraph, a comparison between the R_{on_spec} , total semiconductor losses, junction temperature, heatsink volume, semiconductor active area, current density and power loss density of bulk Diamond devices against GaN on GaN and 4H-SiC power devices is provided. In this comparative analysis, only unipolar devices will be considered. Above 3kV and at 10kV in particular, vertical GaN devices on bulk GaN substrates will be considered for this study. 4H-SiC vertical bulk devices, already demonstrated to withstand up to 27kV [191], will be considered in this analysis.

The models and basic assumptions for diamond devices are based on those used in the previous sections, including incomplete ionization, simplified switching loss model, NPT vertical drift region, doping and temperature dependent mobility, impact ionization coefficients for breakdown voltage. The parasitic output capacitance (C_{oss}) and the ON state resistance are assumed to be due only to the drift region. For the heatsink volume, natural convection is considered with a volumetric resistance of $500\text{cm}^3 \cdot ^\circ\text{K}/\text{W}$ [192].

This value typically overestimates the heatsink volume, whereas forced air solutions can reduce the heatsink volume by a factor of 5 to 10. However, this assumption will allow a quantitative benchmark on diamond devices versus other materials. The ambient temperature will be set to 300K (table 12-13).

For a 10kV breakdown voltage, the drift region of bulk diamond, bulk GaN and 4H-SiC devices are based on the parameters shown in table 11. The switched voltage and current are 6kV and 10A, respectively. The duty cycle is set at 0.5 and the switching frequency at 20kHz (note that this is a very high switching frequency for such high voltage power devices).

The comparison between vertical unipolar Bulk diamond, GaN on GaN substrates and 4H-SiC is presented in table 12-13, figures 24-27. Key elements can be highlighted:

- Even at the “low” temperature of 400K, diamond power devices have 29% less total losses and heatsink volume than GaN, requiring 22% less active area. With diamond, the current density is around 42A/cm², and the loss density is under 130W/cm² at 20kHz, albeit 17% more than GaN. The analysis is similar with 4H-SiC, where diamond has 11% less total losses and heatsink volume, with 21% more current density.
- At 450K, Diamond devices have 33% higher current density, 28% less losses and heatsink volume than 4H-SiC at the same temperature. Diamond devices have 7% higher power loss density than 4H-SiC. Compared to GaN at the same temperature, Diamond devices have 20% higher current density, 19% less losses and heatsink volume.

Increased junction temperature of 500K will allow Diamond devices to gain benefit from the increased dopant activation (no compensation is assumed here). At this junction temperature diamond devices outperform both SiC and GaN at the same junction temperature. The benefits of Diamond devices are even higher if compared with GaN and SiC operating at lower junction temperature. In detail:

- Diamond devices at 500K would require 58% less heatsink volume than GaN operating at 400K, with at least 17% less losses and half active area. The current density in diamond at 500K is approximately 40% higher than GaN at 400K.
- Diamond devices at 500K would require 40% heatsink volume of that of 4H-SiC devices operating at 400K, with 20% less losses and 25% less active area. The current density in diamond at 500K is 32% higher than that of 4H-SiC devices at 400K.

Table 11. Device parameters used for the 10kV comparison.

10kV	Drift region Thickness	Drift region doping	Max. Electric Field	Ron_spec (450K)	Ron_spec (500K)
Bulk Diamond p-type	47μm	2.7e15cm ⁻³	4MV/cm	63.7mOhm.cm ²	60.8mOhm.cm ²
Bulk GaN on GaN n-type	90.5μm	1.1e15cm ⁻³	2MV/cm	114mOhm.cm ²	206mOhm.cm ²
4H SiC n-type	90μm	1.2e15cm ⁻³	2MV/cm	133mOhm.cm ²	171mOhm.cm ²

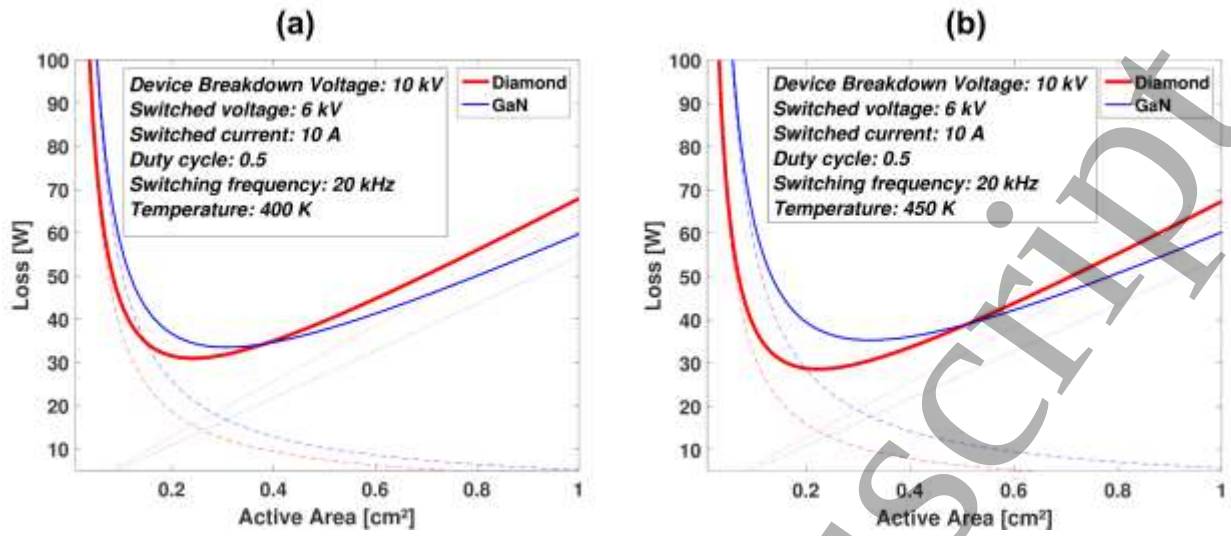


Figure 24, Semiconductor total losses as a function of active area for Bulk diamond and GaN for the 10kV target. a) 400K. b) 450K. Total losses (solid lines) are originated from ON state(dashed) and switching losses(dotted).

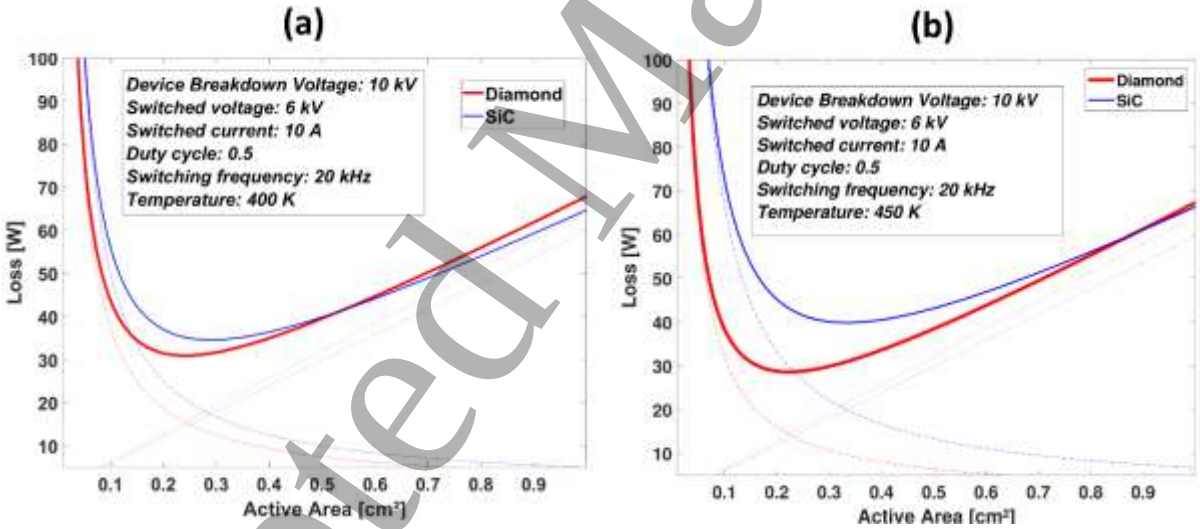


Figure 25, Semiconductor total losses as a function of active area for Bulk diamond and 4H-SiC for the 10kV target. a) 400K. b) 450K. Total losses (solid lines) are originated from ON state(dashed) and switching losses(dotted).

Table 12. Comparative case study between GaN on GaN and bulk diamond for the same specifications.

6kV (BV 10kV) 10A 0.5 duty cycle - 20kHz		Diamond	GaN	Diamond	GaN	Diamond	GaN
Junction Temperature	K	500	500	450	450	400	400
Optimal Area	cm ²	0.22	0.435	0.225	0.325	0.24	0.305
Conduction Loss	W	≈14	≈23.6	≈14	≈17.6	≈15.5	≈16.9
Switching loss	W	≈14	≈23.7	≈14	≈17.7	≈15.5	≈16.6
Total loss	W	28	47.37	28.6	35.3	30.96	33.49
Current density	A/cm ²	45.5	23	44.4	30.8	41.7	32.8
Power loss density	W/cm ²	127	108.9	127	108.6	129	110
Heatsink Volume	cm ³	70	118.4	95.4	117.7	154.8	167.5

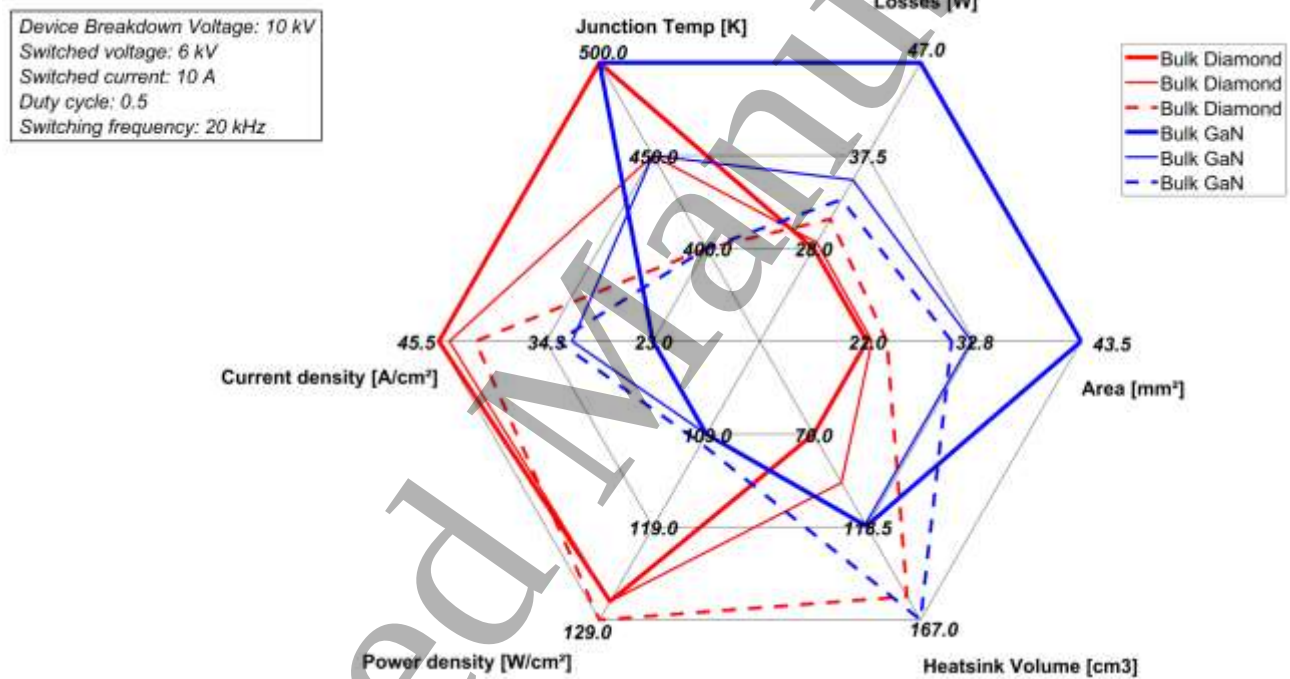


Figure 26, Spider chart comparing GaN and Diamond power devices and their impact at system level.

For the 10kV range, diamond devices are expected to have smaller losses, heatsink volume and active area than other materials. Diamond is still superior to 4H-SiC even at 400K, although the gap in the performance is significantly amplified above 450K.

Table 13. Comparative case study between 4H-SiC and bulk diamond for the 10kV comparison.

6kV (BV 10kV) 10A 0.5 duty cycle - 20kHz		Diamond	4H-SiC	Diamond	4H-SiC	Diamond	4H-SiC
Junction Temperature	K	500	500	450	450	400	400
Optimal Area	cm ²	0.22	0.38	0.225	0.335	0.24	0.29
Conduction Loss	W	≈14	≈22.5	≈14	≈19.9	≈15.5	≈17.3
Switching loss	W	≈14	≈22.7	≈14	≈20	≈15.5	≈17.3
Total loss	W	28	45.22	28.6	39.86	30.96	34.61
Current density	A/cm ²	45.5	26.3	44.4	29.9	41.7	34.5
Power loss density	W/cm ²	127	119	127	119	129	119
Heatsink Volume	cm ³	70	113	95.4	132.9	154.8	173

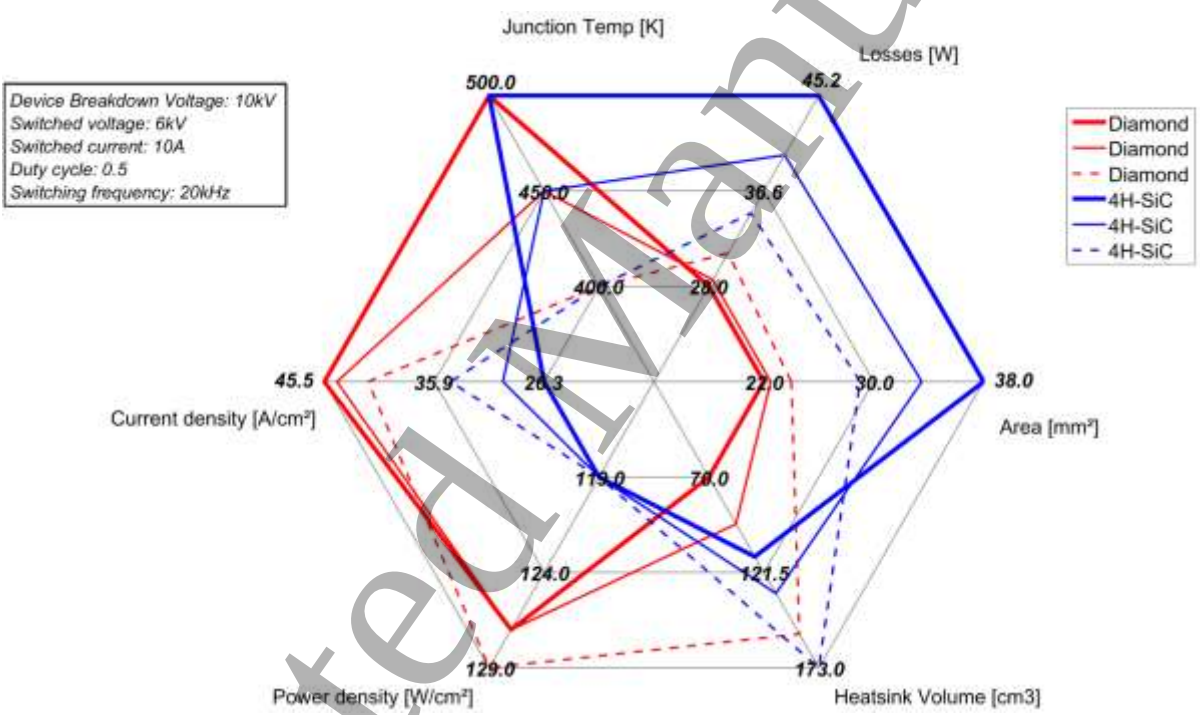


Figure 27, Spider chart comparing 4H-SiC and Diamond power devices and their impact at system level.

It is also possible that the performances of diamond devices are yet underestimated. Indeed, improvements in the crystal structure could result in larger critical electric fields (above what has currently been measured). In that case the impact ionization coefficients would need to be updated particularly if thick and low doping drift regions are considered for the 10kV+ range of rated breakdowns. To some extent this is also the case for bulk GaN above 3kV and 400K, though it is less likely to see a significant difference.

If one would like to compare power devices at an operating temperature close to RT, table 14 shows the system level performances. Please note that industrial ambient temperatures can be as high as

358K, and even up to 398K (e.g. automotive applications). Hence, table 14 also shows the comparison between diamond, SiC and GaN, for the same 10K difference between the junction temperature and the ambient temperature, but with an ambient temperature of 400K. By comparing tables 12-14 one can also note the impact of lower junction temperatures on the heatsink volume and the total losses.

As shown in table 14, bulk diamond devices are not well suited for an operation at $T_j \approx 300\text{K}$. This is no longer the case when the ambient temperature is increased. Moreover, if a large temperature gradient between the junction temperature and the room temperature of 300K is desired to reduce the heatsink volume or to increase the current rating, then diamond has significant system-level benefits, as presented in tables 12 and 13.

Table 14. Comparative case study between 4H-SiC, bulk GaN and bulk diamond for a 10K temperature difference between the junction temperature and room temperature, or an ambient temperature at 400K. Please note that the unit of the heatsink volume has been modified to dm^3 , compared to the previous tables (cm^3).

6kV (BV 10kV) 10A 0.5 duty cycle - 20kHz		Diamond	GaN	4H-SiC	Diamond	GaN	4H-SiC
Ambient temperature	K	300	300	300	400	400	400
Junction temperature	K	310	310	310	410	410	410
Optimal Area	cm^2	0.345	0.245	0.215	0.235	0.305	0.3
Conduction Loss	W	22	13	13	15	17	18
Switching loss	W	22	13	13	15	17	18
Total loss	W	44	26	25	30	34	36
Current density	A/cm^2	29	41	47	43	33	33
Power loss density	W/cm^2	128	108	119	129	110	119
Heatsink Volume	dm^3	2.2	1.3	1.3	1.5	1.7	1.8

6. Future perspective of diamond power devices

6.1. The roadmap of diamond

The main challenges for diamond power devices can be organised in five categories: (i) material, (ii) devices, (iii) packaging, (iv) reliability, and (v) integration. Their current status and future prospective are shown in table 15.

Table 15. Current status and challenges for diamond devices in power electronics.

	Challenge	Current Status	Breakthroughs and future prospective
Material	Wafer size	< 1 inch	> 2 inches
	Cost	> 400\$ per <1inch wafer	Dependent on the BV (see discussion below (i))
	Defects	High dislocation density	Lower dislocation density for increasing wafer size
	Interface quality	Medium-High interface states and defects density compared to Si devices	Optimized annealing techniques for improving the quality
	Doping	p-type with Boron (deep acceptor level), lack of reliable n-type	Lower activation energy dopant species, new conduction mechanisms
Devices	High BV and current FETs	~2kV for lateral technologies	>10kV, >10A for vertical technologies, high transconductance, low threshold voltage, low Ron_spec transistors.
	High Power p-type SBDs	>1kV, >5A	Improve the BV (field terminations, thick and high-quality drift region, etc.) without affecting the ON state current (target >10kV, >10A).
	N-type FET	Not available	Development of techniques for low resistive n-type layers.
	Leakage current	- Dominated by defects - Limiting factor for high BV	Improve material quality
	Lifetime	<10ns	>100ns
	Termination	Not optimized and may lead to TDDb	New solution for high voltage passivation
	Fast switching	Limited to diodes	>100V/ns
	Novel device structures	N/A	SuperJunctions, floating islands, new techniques
Packaging	High temperature packaging	N/A	Unique packaging technique for high temperature operation (i.e. 175°C for 10kV, >200°C for 3kV)
	Ultra-high voltage packaging	N/A	New passivation methods, secondary passivation techniques, etc.
Reliability	Ageing	No ageing test have been conducted so far	Lifetime of diamond devices is still an open issue
	Yield	N/A	Improve the repeatability and reproducibility of devices
	Harsh environment	Only few tests have been performed	Show suitability to harsh environments with more standard tests
	High switching	N/A	Resistance to high di/dt and dv/dt. New switching model. EMI/EMC would need specific filters design.
Integration	Passive components	N/A	On-chip Integrated capacitances ,resistances ,inductances to reduce parasitic
	Active devices	Limited to logic devices	Isolated transistors and/or diodes
	Integrated gate driver	N/A	Smart gate driving for p-type FETs would improve the switching frequency

(i) *Material*: The decrease in the defects and dislocation density together with the enhancement of the interface quality would also enable the reduction of leakage currents (detrimental for high temperature operation) and achieve breakdown voltage levels closer to their theoretical predictions. The high-cost of the substrate is another crucial aspect to take into account for the commercialisation of diamond power devices. Let's consider diamond devices for a 1.2kV application fabricated on a 25cm² (~2.2inch) substrate and recall the price per Ampere (including processing cost) for Si IGBT and SiC MOSFET technologies (0.11\$/A and 0.65\$/A, respectively [193]). Assuming a yield of 60% and an active area which occupies 90% of the total die area (the remaining 10% is for the termination region), it is possible to estimate the equivalent cost of a diamond substrate in order to be competitive with Si IGBT and SiC MOSFET (table 16). A similar calculation has been also made for a ~6kV target (the price per ampere for Si Thyristor has been fixed at 1.5\$/A[194] and 4\$/A for Si IGBT[195]). The ON state current and current density, optimal active area, junction temperature and switching frequency have been for diamond devices have been taken from tables 4 and 12. Based on these assumptions, the equivalent cost of a 2.2-inch diamond substrate for a 1.2kV application which is competitive with SiC MOSFET and Si IGBT is calculated to be ~900\$ and ~150\$; for 6kV the substrate cost needs to be ~450\$ in order to be competitive with Si Thyristor or ~1200\$ comparatively to the Si-IGBT. Nonetheless, it is worth mentioning that as Si Thyristor are more suited for ultra-low switching frequency (<1kHz) and the comparison with Si-IGBT is more appropriate for higher switching frequency. Higher current densities allowed by, as an example, increased junction temperature, reduced ionization energy or a higher loss density will allow to increase the equivalent cost of a diamond device (up to ~4400\$ to be still competitive with SiC for 1.7kV breakdown voltage). Consequently, the reduction of the specific ON state resistance more than the decrease of the total cost and the increase of the total area of each diamond wafer, appears to be the best strategy to enable the commercialization of diamond.

Table 16. Equivalent cost for 2.2inch diamond substrate for different application. The suggested (equivalent) diamond device is able to switch the target voltage for a 0.5 duty cycle at f=20kHz.

hp: 0.5 duty cycle f=20kHz	Diamond, T=450K, 1.2kV				Diamond, T=500K, 6kV			
ON state current (A)	50	250			10	22		
ON state current density (A/cm ²)	200	1000			45	100		
active area (cm ²)	0.25	0.25			0.22	0.22		
die area (cm ²)	0.28	0.28			0.24	0.24		
wafer size (cm ²)	25	25			25	25		
yield	0.6	0.6			0.6	0.6		
total number of dies	90	90			102	102		
number of good dies	54	54			61	61		
gross margin	50%	50%			50%	50%		
comparison	SiC MOSFET	Si IGBT	SiC MOSFET	Si IGBT	Si Thyristor	Si IGBT	Si Thyristor	Si IGBT
cost per chip (\$)	16.25	2.75	81.25	13.75	5	20	11	44
price per chip (\$)	32.5	5.5	162.5	27.5	10	40	22	88
cost per ampere (\$ (FIXED)	0.65	0.11	0.65	0.11	1.5	4	1.5	4
cost of diamond 2.2inch substrate (\$)	877.5	148.5	4387.5	742.5	458	1220	1006	2684

(ii) *Devices*: Technological progress in the wafer and processing quality will also have to focus on four key aspects related to device technology and performance: (a) the development of vertical devices and novel structures, (b) carrier lifetime control, (c) device termination optimisation which could benefit from the fabrication of multi-layer passivation based on high-k materials, and (d) enhanced slew rate. In addition, the demonstration of a high-current diamond p-type FET and high-current high-voltage n-type FET would also lead to smart integration of key components, such as protection, drive and possibly R,L,C passives within the same chip. Vertical HFETs and depletion mode FETs seem the most promising solution for high current-voltage p-type devices.

(iii) **Packaging:** Packaging of diamond devices is an area where significant development will be required in particular to accommodate the very high slew rates or the operation in harsh environment applications. It is therefore apparent that unique packaging technique, not compatible with those available for other WBG materials, need to be put forward.

(iv) **Reliability:** As a further step towards the commercialization, diamond devices will need to pass reliability tests which can guarantee the device performance over a specified time period and give an estimation of device lifetime in standard and harsh environment conditions. It is very likely that as the development of diamond-based devices progresses, dedicated standards and reliability tests will be defined for diamond.

(v) *Integration:* Wafer integration of parasitic and smart gate driving circuits will enable low-volume diamond converters and improve dV/dt and the dI/dt for the packaged devices. The smart integration of diamond and other WBG/UWBG semiconductors (figure 28) could represent one of the possible applications of diamond p-type FETs in a monolithic high-speed converter.

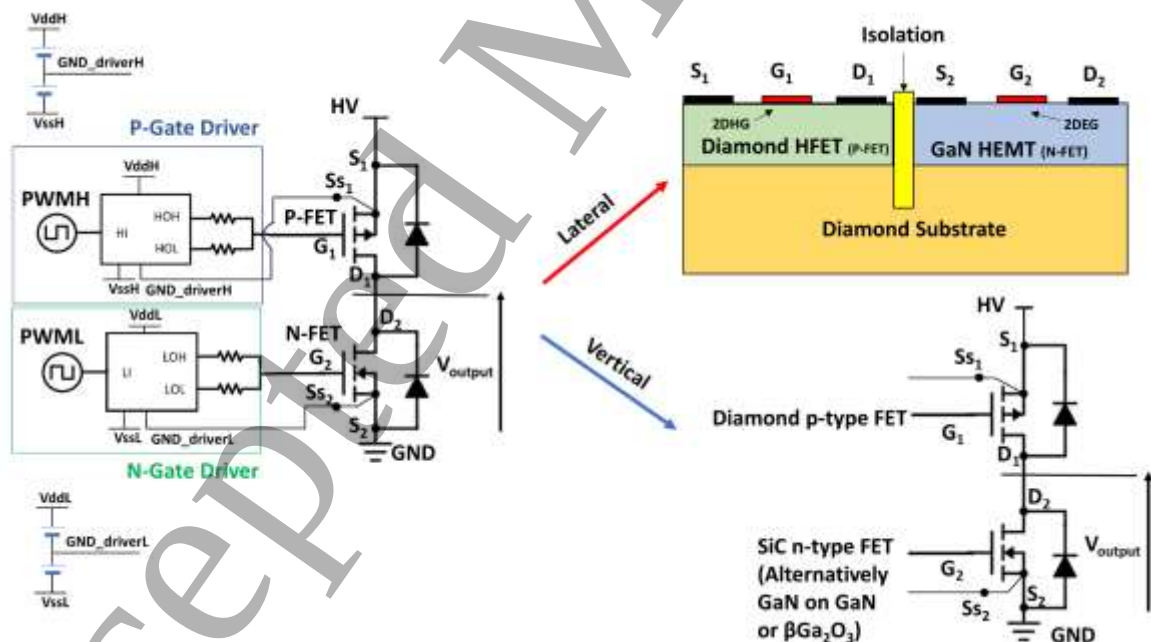


Figure 28, schematic of a possible monolithic implementation of diamond and other WBG/UWBG semiconductors. The integration for the reduction of parasitics and better thermal performance (through the common diamond substrate). Kelvin sources have been added in view of the fast switching speed achievable by using such a configuration.

6.2. Conclusions

The remarkable advantages of diamond for power electronics have been accompanied by drawbacks and limiting factors. While some of the challenges which have hampered for years the development of this material have been addressed, other questions still remain unanswered. With the principal aim of reviewing the diamond technology and modelling the most promising devices, this topical review has provided a new thorough overview of the current state-of-the-art and future trends. Significant efforts, especially in the last decade, have resulted in a class of new diamond power devices which have the potential to find their place in the diverse power electronics market in the future. The comprehensive analysis conducted in this paper has however demonstrated that the superior potential of diamond is mainly restricted for high junction temperatures ($>450\text{K}$), medium-high frequency ($>20\text{kHz}$) and high voltage ($>3\text{kV}$) applications when compared to 4H-SiC and GaN commercial alternatives. Besides, the results illustrated in this work have pointed out the significant thermal management issues which needs to be addressed to allow for efficient parallelization of multiple diamond devices with a negative temperature coefficient.

Acknowledgements

The authors would like to thank Cédric Masante, Gaetan Perez and Aurélien Marechal for their contributions to the modelling of diamond devices presented in this article.

This work was supported by the U.K. Engineering and Physical Sciences Research Council for the University of Cambridge Centre for Doctoral Training under Grant EP/M506485/1 and by the French ANR Research Agency under grant ANR-16-CE05-0023 #Diamond-HVDC. The research leading to these results has been performed within the GREENDIAMOND project and received funding from the European Community's Horizon 2020 Programme (H2020/2014–2020) under grant agreement no. 640947.

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